Microstructure and Properties of Hastelloy X Fabricated by Additive Manufacturing

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Many Exciting Research Topics on Additive Manufacturing





- Microstructure control
- Modeling at various stages
- In-situ monitoring
- Data Analytics



Project Objectives

- Optimize additive manufacturing (AM) fabrication processes for solution strengthened Hastelloy X (HX, Ni-22Cr-19Fe-9Mo) gas turbine components (Fuel injector, Combustor)
- Properties requirement depends on component and application (prototyping, production and repair)
- Compare two AM techniques, electron beam melting (EBM) and selective laser melting (SLM).
- Generate data (Tensile, Fatigue, Creep, Oxidation) relevant for FE applications for Hastelloy (HX, Ni-22Cr-18Fe-9Mo) alloy
- Effect of annealing or HIP'ing on microstructure and mechanical properties
- Effect of different EBM precursor powders on mechanical and oxidation behaviors



HX Made by EBM and SLM



Small SLM HX Cube for Parameter Optimization Based on Microstructure



- Spot time
- Spacing
- Energy



ORNL SLM Builds For Microstructure and Mechanical Characterization



- Previous build fabricated with leftover powder from AMO project
- Characterization of the microstructure and tensile properties in 3 directions
- Presentation on vertical specimens
- Study the effect of annealing and Hip'ing



New SLM (SLM-Opt) Build For Extensive HX Characterization



- 35h, ~2000 layers
- 65mm tall
- Rectangular blocks to study properties anisotropy
- Thin wall effect
- Machine available for 2 weeks for process optimization



Fabrication of 20-30 EBM Specimens For Tensile, Creep and Fatigue Testing



- 27h, 1240 layers, ~65mm
- Small builds first to optimize parameters based on 718 & HX previous work
- Similar build parameters + pre-heat temperature
- Study the effect of post annealing and Hip'ing



Similar Composition For EBM, SLM and Wrought HX Except for Si &Mn

Solution Strengthened Hastelloy X (Ni-22Cr-19Fe-9Mo)

	Ni	Cr	Fe	Мо	Со	Mn	Si	W	С
EBM Powder	Bal.	21.76	18.43	8.91	1.51	0.07	0.08	0.6	0.08
EBM Alloy	Bal.	21.38	18.55	9.05	1.55	0.01	0.05	0.64	0.078
EBM(Si,Mn) Pow.	Bal.	21.7	18.7	9	1.56	0.93	0.86	0.66	0.06
EBM(Si,Mn) Alloy	Bal.	21.43	18.87	9	1.56	0.67	0.71	0.65	0.048
SLM Powder	Bal.	21.47	18.83	8.96	1.51	0.01	0.16	0.63	0.07
SLM-Opt Powder	Bal.	21.72	18.51	8.87	1.51	0.01	0.06	0.6	0.08
Wrought	Bal.	22.06	17.86	9.53	1.8	0.65	0.31	0.6	0.067

- Alloy composition consistent with EBM powder composition
- High concentration of Mn and Si in EBM(Si,Mn) and wrought HX. Specification: Mn and Si <1%



EBM/EBM(Si,Mn) Powder Morphology Typical of Gas Atomized Powder



- Most powder particles contain large numbers of satellite particles
- Irregularly shaped particles



Powder Defect: Larger Voids for the EBM powder





Wrought HX: Random Distribution of (Mo,Si) Precipitates



EBM: Larger Grains at the Center vs Surface



Fabrication of contour first







Elongated Grain Along Z Direction More Precipitates for EB(Si,Mn)

EBM(Si,Mn) = Larger grains

EBM = Larger voids



EBM(Si,Mn): (Mo,Si)-rich Carbides EBM: (Mo,Cr)-rich Carbides at GB

EBM(Si,Mn)

EBM





SLM-Opt: Small Grains, No Precipitate Hot Tearing Cracks





Lower defect density for SLM-Opt Consistent with literature Defect could likely be eliminated

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Both EBM & EBM(Si,Mn) HX **Alloys Exhibit Good Ductility But Lower** Strength < 800°C



400

300

200 -

100.

Cast HX

Min Req.

EBM(Si,Mn)

Havnes data IX sheet

EBM

SLM HX Exhibits High Strength but Moderate Ductility



- High YS due to high residual stress
- Higher ductility for SLM-Opt



EBM & EBM(Si,Mn): Good Low-Cycle Fatigue Properties at 800°F/425°C



- Fully-reversed LCF
 - Consistent with excellent HX EBM alloys ductility
 - Similar results at other temperatures



EBM&EBM(Si,Mn): Lower Creep Strength & Higher Ductility at 816°C Similar Results for the Two Alloys



SLM: Similar Creep Strength but Limited Ductility at 816°C





EBM: Fully Dense Material after HIP'ing at 1177°C/2h/150MPa, "Fast Cooling"



HIP'ed EBM: Increasing Number of Small Grains + Precipitates Along Few Grain Boundaries









EBM(Si,Mn): Fully Dense Material after HIP'ing at 1177°C/2h/150MPa



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HIP'ed 2h/1177ºC/150MPa <u>50 μm</u>

Main microstructure differences between as fab, annealed and HIP'ed HX are voids and precipitates in the grains





EBM(Si,Mn): No Clear Effect of Hip'ing on the Creep Properties at 816°C Improvement for EBM?



EBM,750°C: Better Creep Results For Short Term Tests?

Significant improvement after HIP'ing for EBM alloy





SLM: Fully Dense Material After HIP'ing at 1177°C/2h/150MPa+Recrystallization



SLM: Increase of Ductility & Decrease of YS after 15min 1177°C or HIP'ing 2h/1177°C/150MPa

Release of Residual Stress + Microstructure evolution





SLM: Result Consistent with Literature Data at Room Temperature



SLM: Significant Improvement of Creep Deformation at 816°C After Hip'ing Consistent with Tensile Data



SLM: Long Term Creep Tests Have Been Running for 2,300h at 750°C



 Higher deformation during primary stage for HIP'ed HX
 110MPa tests

coming soon



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Deformation (%)

Need Oxidation/Corrosion Studies on AM Ni-based Materials

- Different microstructure & surface finish = various oxidation behaviors?
- Topology optimization using AM will lead to thinner components and lower oxidation lifetime
- Optimization of the microstructure at the surface for better oxidation/corrosion behavior?



100h TGA Tests at 950°C in Air to Assess AM HX Oxidation Resistance

EBM

100



100h TGA at 950°C : Spallation Initiates at Grain Boundaries for EBM(Si,Mn)



Cyclic Oxidation Testing of HX in Box Furnace at 950°C, 100h Cycles



- 950°C for fast assessment of long-term oxidation behavior
- As Fabricated AM coupons. Polished & not polished
- Crucible to generate both specimen and gross mass gains
- Gross mass gains = oxygen pickup ~ Metal (Cr) loss
- Surface imaging using Keyence 3D microscope



Faster Initial Oxidation Rate for EBM(Si,Mn) and 2 EBM specimens



- Similar Gross MG for Wrought & one EBM specimen
- Lower GMG for SLM and SLM-Opt specimen



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High Spallation for EBM(Si,Mn). Moderate Spallation for SLM & EBM. Increasing Spallation for Wrought HX



- Change of spallation rate with time
- Evaporation likely negligible compare to spallation



Decrease of Spalled Area Fraction With Time for AM Specimens. Increase for Wrought HX



Decrease of Spalled Area Fraction With Time for AM Specimens. Increase for Wrought HX



Image Analysis & Mass Change Data/Modeling Yield Similar Spallation Fraction



10x100h Cycles: Voids formation in the Grain and at GB for EBM HX



1000h, 950°C, Wrought & EBM(Si,Mn): (Cr,Mn) and Cr-Rich Layers

Wrought

EB(Si,Mn)





1000h, 950°C, Wrought and EBM(Si,Mn): Si segregation at Interface + Oxide Penetration along GB



Cr-rich oxide + numerous voids at GB Mo depleted zone for EBM(Si,Mn)



Locally Less spallation after 15min 1177°C Fast Cooling for EB(Si,Mn)



Conclusion-EBM

- EBM HX shows good ductility and tensile strength superior to the cast HX requirement
- Good fatigue properties for the EBM HX but lower creep strength. Sufficient for some applications using cast HX?
- EBM & EBM(Si,Mn)HX microstructure could certainly be further optimized to increase the alloy creep performance
- Hip'ing resulted in a significant improvement of HX EBM creep properties. No effect on EBM(Si,Mn).
- Significant effect of Si & Mn content on microstructure and oxidation.
- Great opportunity (already happening) to develop new AM HX alloy



Conclusion-SLM

- Hot tearing led to crack formation as observed by others.
 Need to optimize fabrication parameters or alloy composition
- SLM HX exhibited good tensile strength and acceptable ductility. Good creep strength but limited ductility and high residual stress
- HT or Hip'ing increased the SLM HX ductility & reduced YS. Creep ductility was significantly improved by Hip'ing. Similar properties as wrought HX
- Increased spallation was observed for SLM HX alloy after 100h at 950°C but lower spallation rate after 3,000h.
- Will start soon to generate data perpendicular the build direction



