# Corrosion Issues in Advanced Coal-Fired Boilers

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# Acknowledgments

#### ORNL

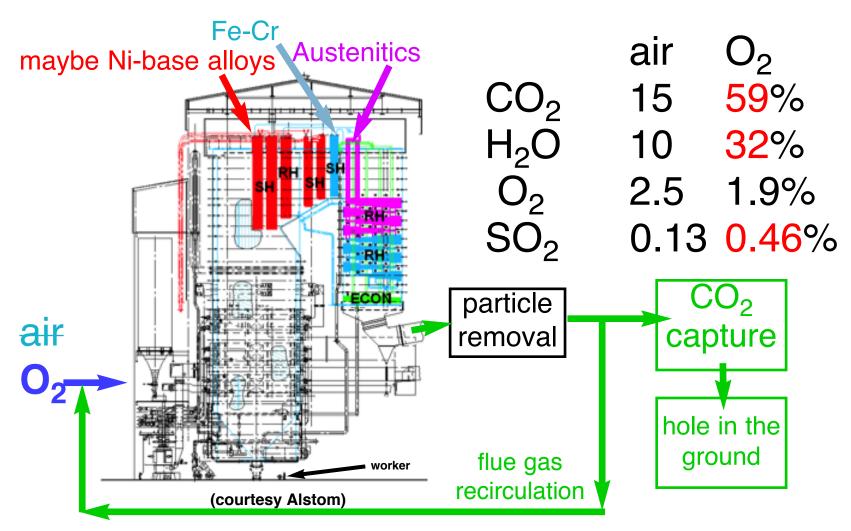
- G. Garner, T. Lowe, M. Stephens, M. Howell oxidation experiments
- J. Moser, T. Jordan, A. Willoughby, D. McClurg creep testing
- T. Jordan metallography
- T. Lowe SEM, image analysis
- D. Leonard EPMA, image analysis

### FY10-13 Tasks & Timeline

- Goal: Mechanistic understanding to enable accurate oxy-fired corrosion modeling
- 1. Steam/gas corrosion (no ash)
- 2. Fireside corrosion (with ash)
- 3. Environment-mechanical property effect
  - effect of steam on creep (Dryepondt)
- A. ~600°C ferritic/martensitic steels (FY10-12)
  - creep testing at 650°C (FY12-13)
- B. ~650°-700°C austenitic steels (FY11-13)
- C.~700°-750+°C Ni-base alloys
  - creep testing at 800°C (FY11-12)
  - ash testing 600°-800°C (FY12-13)

#### Advanced: Oxy-firing to facilitate CO<sub>2</sub> C+S

Retrofit current plants or advanced 760°C (below)



Several studies published by Alstom (Bordenet)

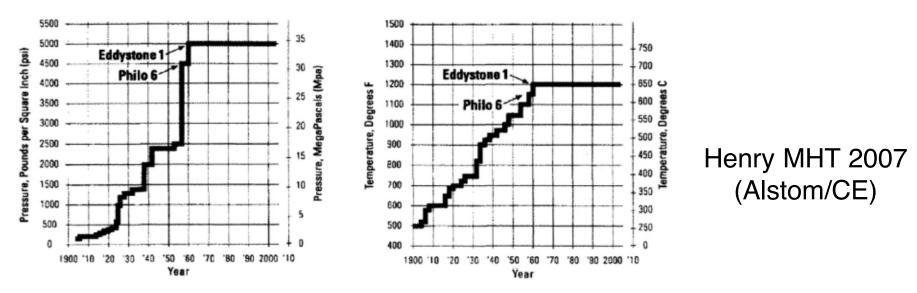
What is the effect of oxy-firing on fireside corrosion?

#### Ultimate goal is to marry Oxy + A-USC

"least regret" CO<sub>2</sub> strategy: higher efficiency

A-USC: 760°C (1400°F) + 34.5 MPa (5000psi)

(Advanced ultra-supercritical)



History: 1960 - the year progress stood still

Eddystone (1960): 654°C/36.5MPa (1210°F/5300psi) settled for 613°C/34.5MPa (1135°F/5000psi)

Turk (2013): 599°/607°C SH/RH 25.3MPa (1110/1125F)

# Corrosion testing with ash

Determine effect of Temp., CO<sub>2</sub>, H<sub>2</sub>O, SO<sub>2</sub>...



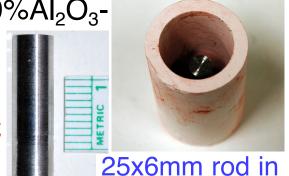
Synthetic ash: 30%Fe<sub>2</sub>O<sub>3</sub>-30%Al<sub>2</sub>O<sub>3</sub>-

30%SiO<sub>2</sub>-5%Na<sub>2</sub>SO<sub>4</sub>-5%K<sub>2</sub>SO<sub>4</sub>

Gas:  $N_2$ - $CO_2$ - $H_2$ O- $O_2$ - $SO_2$ 

Temperature: 600°-800°C

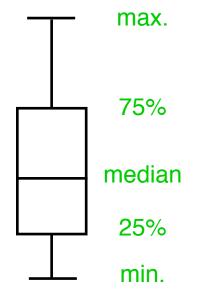
Time: 500 h

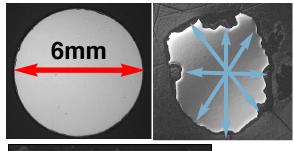


25x6mm rod in porous alumina

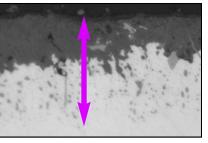
#### Box and whisker plots

oxide thickness or metal loss (128)





remaining metal (radii)



oxide thickness = scale + internal oxide

#### Range of commercial & model alloys

measured by inductively coupled plasma analysis and combustion analysis

#### Alloy chemical compositions (weight %) Alloy UNS# Cr Ni Mo W Mn Ν Other Fe 95.5 2.3 0.2 0.9 Gr.22 K21590 0.6 0.1 0.2Cu 0.14 0.01 < Gr.33 0.1 0.9 0.01 0.3 0.3V.0.07Nb Gr.91 S90901 8.3 0.1 0.08 0.05 SAVE12 83.4 9.6 0.3 0.05 3.0 0.1 0.11 0.01 2.6Co,0.3V 0.4 304H 69.7 18.9 8.5 0.3 0.04 1.0 0.05 0.02 0.3Cu,0.1Co S30409 1.1 Super304H S30410 68.0 19.0 8.9 0.1 < 0.4 0.1 0.08 0.11 2.9Cu,0.1Co 66.0 18.6 11.8 0.2 0.02 1.5 347HFG 0.4 0.09 0.06 0.8Nb,0.2Co,0.2Cu 310HCbN S31042 51.4 25.5 20.3 0.1 0.01 0.3 0.05 0.27 0.4Nb 1.2 43.2 19.7 33.8 0.2 0.02 1.0 0.3 0.08 0.01 0.7AI,0.5Ti,0.3Cu 800H SAVE25 51.5 22.3 20.0 0.1 0.2 0.07 0.22 1.0 0.7 3.4Cu.0.3Nb.0.2Co SANICRO25 42.6 22.5 25.4 0.2 3.4 0.5 0.2 0.06 0.21 2.9Cu,0.5Nb,1.4Co HR120 35.0 24.7 37.6 0.3 0.05 0.7 0.2 0.06 0.21 0.6Nb,0.2Cu,0.1Al 6.3 0.2 23.3 23.4 44.6 0.2 HR6W 1.0 0.07 0.01 0.4Co.0.2Nb.0.1Ti 740 N07740 23.4 48.2 0.5 0.3 0.5 0.08 0.01 20Co,2Nb,2Ti < 0.6 21.6 55.9 8.6 0.09 0.02 0.12 0.05 0.01 617(CCA) N06617 11Co.1.3Al.0.4Ti 282 N07208 0.2 19.3 58.0 8.3 0.1 0.1 0.06 0.01 10Co,1.5Al,2.2Ti < Fe-15Cr 85.1 14.8 < < < < < Fe-20Cr 80.3 19.7 < < 0.01 < < < < 74.6 25.3 0.02 Fe-25Cr < < 0.01 < < < 69.7 30.2 0.02 Fe-30Cr < < < < < < Fe-40Cr 59.6 40.2 0.01 < 0.09 < additions of 0-2%Al, 0-2%Ti, 0-20%Co, 0-8%Mo Ni-(18-22)Cr

<sup>&</sup>lt; indicates below the detectability limit of <0.01%

# Oxy-firing is complicated

Bordenet (Alstom) published worst case

- hot flue gas recycle with no cleaning (unlikely, flue gas cools oxy-fired boiler)
- options: cooling flue gas + de-sulfurization

	air-fire	O <sub>2</sub> -fire	(w/Flue Gas De-S)	
		(hot)	(warm)	(cold)
$CO_2$	15	61%	61%	83%
$H_2O$	10	30%	30%	10%
$O_2$	3	3%	3%	3%
$SO_2$	0.15	0.45%	0.15%	0.15%

#### 500h: Oxy not worse

"air-firing" open box (128 radii)
"oxy-firing" shaded box

"Oxy-firing" did not consistently increase metal loss

SAVE12+304H baseline data

Most protective: 310HCbN, 740 and Fe-30Cr do not show peak at 700°C:

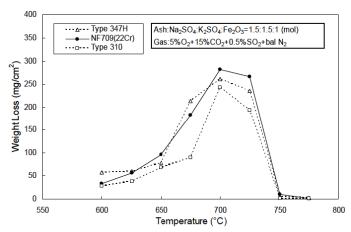
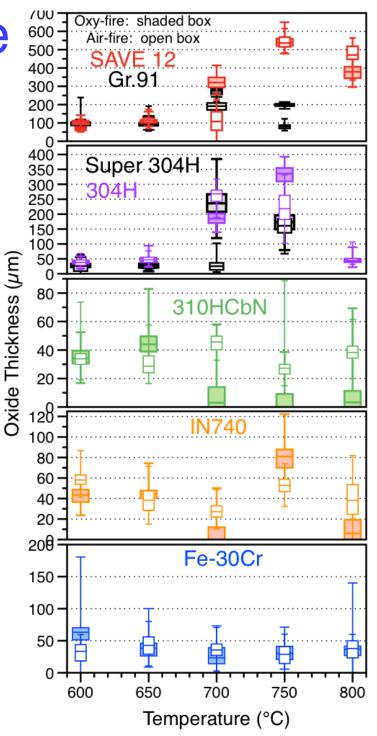
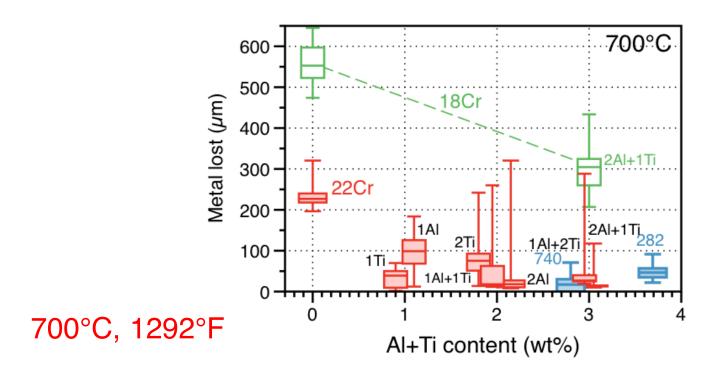


Fig. 3-7 Hot corrosion of NF709 in simulated coal ash environment. (Test duration: 100h)



#### Model NiCr+Al,Ti: Cr key, Ti,Al good

Coupons: 500h in synthetic coal ash + "oxy-firing" gas



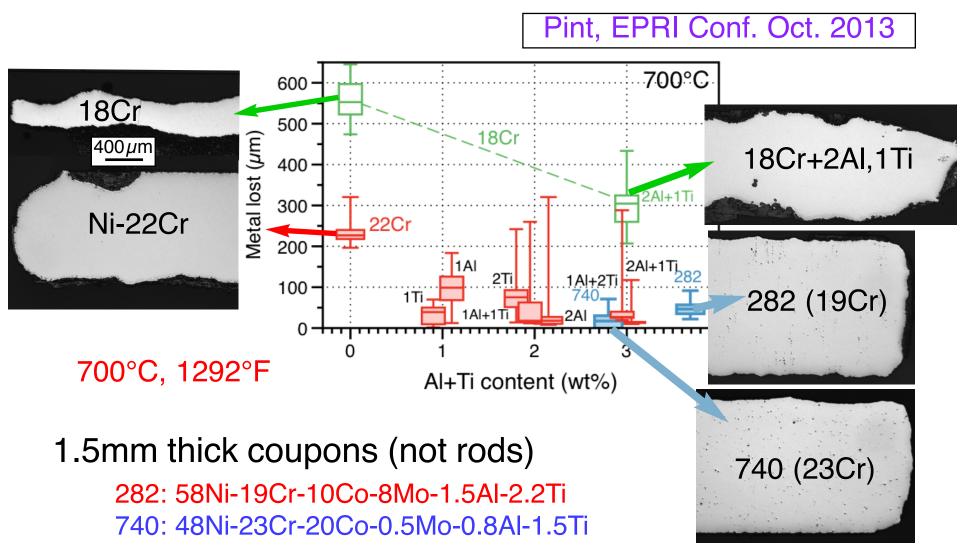
#### 1.5mm thick coupons (not rods)

282: 58Ni-19Cr-10Co-8Mo-1.5Al-2.2Ti 740: 48Ni-23Cr-20Co-0.5Mo-0.8Al-1.5Ti

Al and Ti additions do not explain 740/282 performance

#### Model NiCr+Al,Ti: more sensitive to Cr

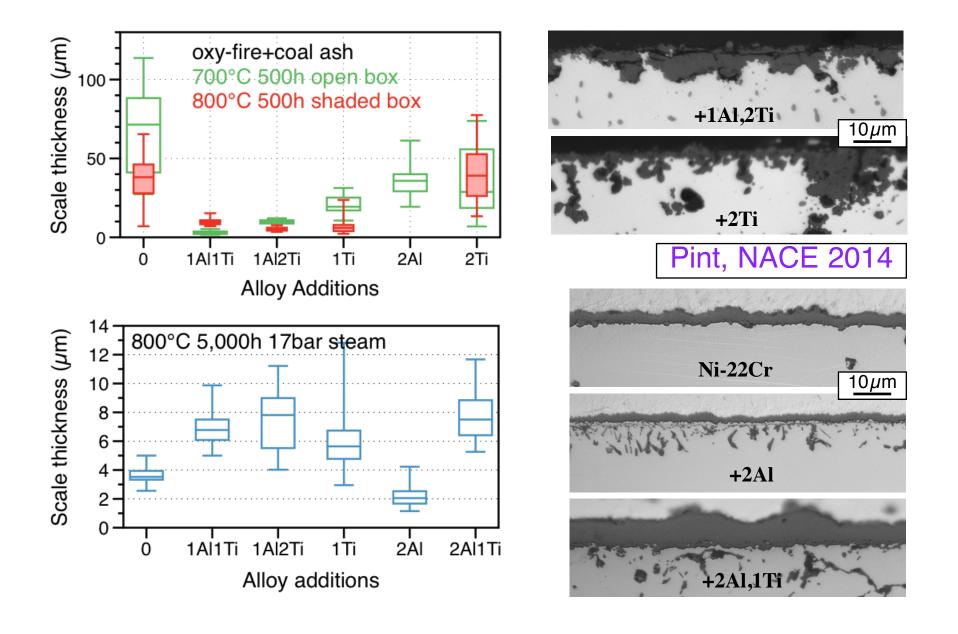
Coupons: 500h in synthetic coal ash + "oxy-firing" gas



Al and Ti additions do not explain 740/282 performance

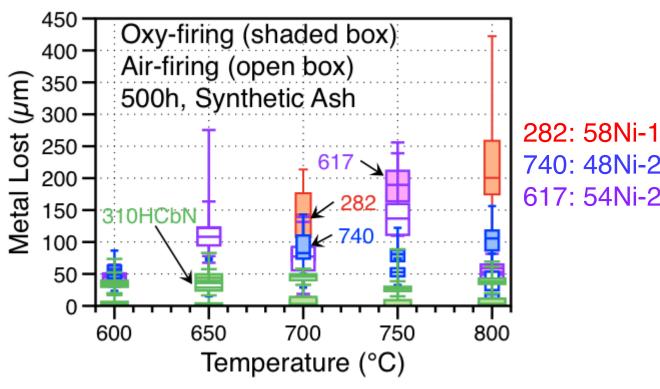
#### Ti: thicker scale in steam

#### Barrier layer prevents direct metal-ash contact



#### Range of behaviors for Ni-base alloys

But none as good as Fe-base 310 stainless steel



282: 58Ni-19Cr-10Co-8Mo

740: 48Ni-23Cr-20Co-0.5Mo

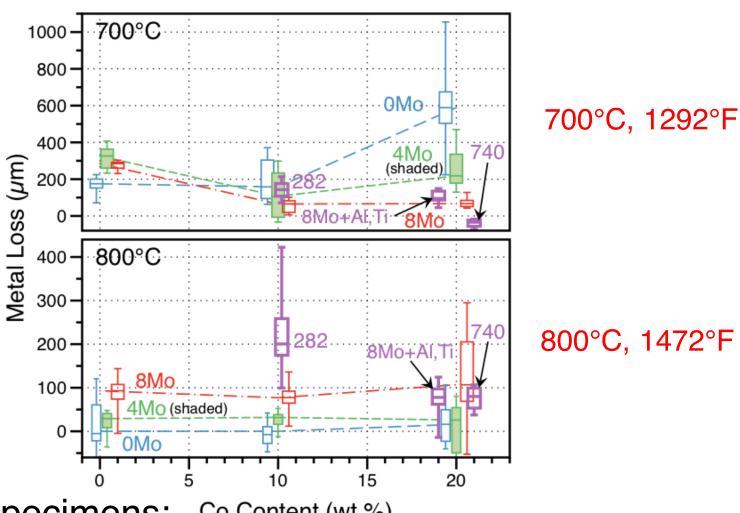
617: 54Ni-22Cr-10Co-9.2Mo

All rod specimens
If Al and Ti are not important:

How much do Co and Mo affect performance?

#### Model NiCrCoMo alloys: muddled

Rods: 500h in synthetic coal ash + "oxy-firing" gas

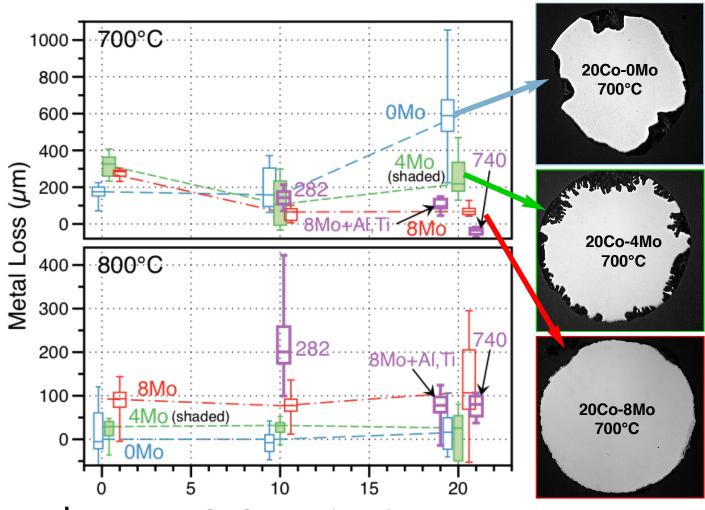


Rod specimens: Co Content (wt.%)

Ni-20Cr-(0-20)Co-(0-8)Mo

#### Model NiCrCoMo alloys: muddled

Rods: 500h in synthetic coal ash + "oxy-firing" gas

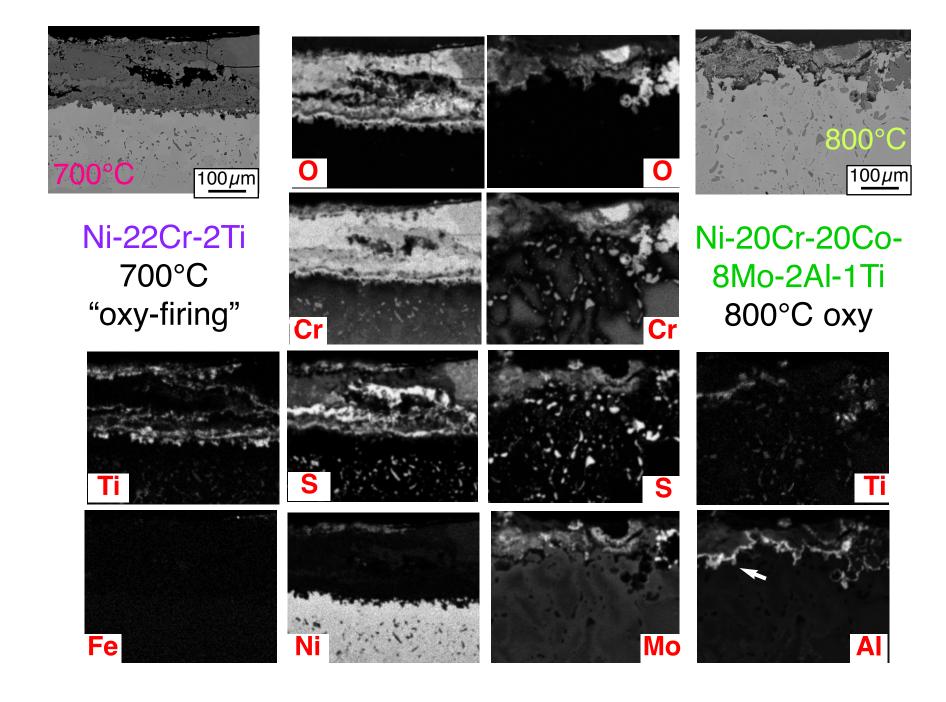


Rod specimens: Co Content (wt.%)

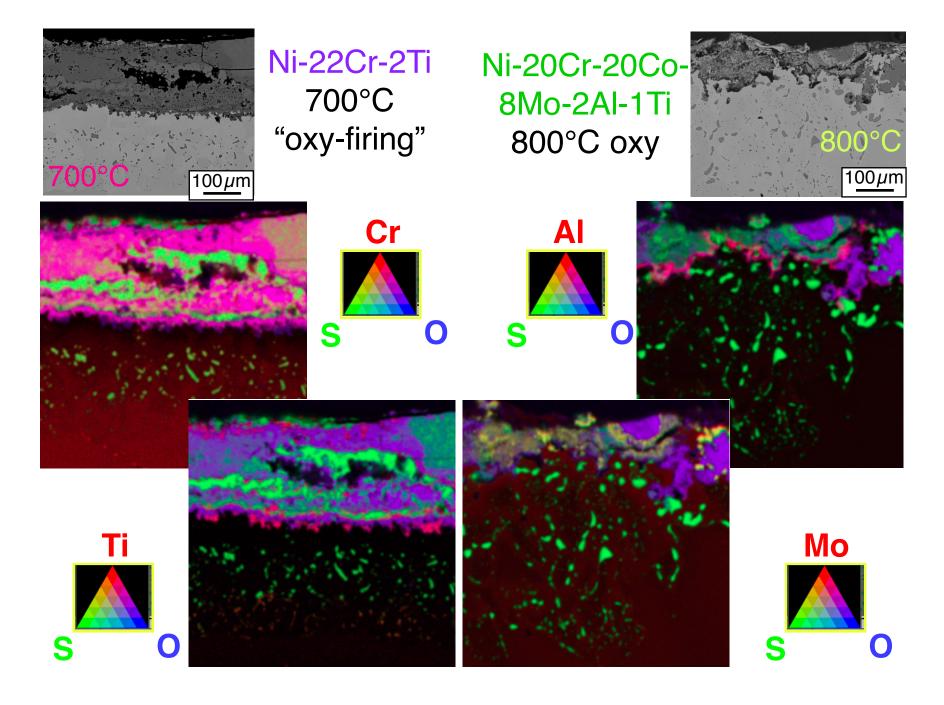
Ni-20Cr-(0-20)Co-(0-8)Mo

Pint, ASME 2014

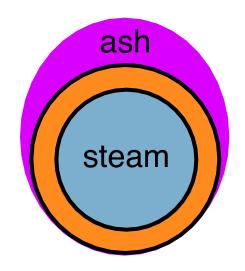
#### Ni-Cr+ash EPMA: Ti,Al in scale



#### Ni-Cr+ash EPMA: Ti,Al in scale



# Boiler tubes see two harsh environments



Expect steam-side oxide to grow by OH<sup>-</sup> transport Any concern about H affecting alloy properties?

# Summary: Creep in steam

1. 800°C (completed): Dryepondt, et al. Mater. Corr. 2012 740 (Ni-23Cr-20Co); 230 (Ni-22Cr-14W) air vs. steam in-situ vs. ex-situ anneal (thermal effect) microstructure analysis

2. 650°C: Grade 91 (9Cr-1Mo) air vs. steam role of thick scale

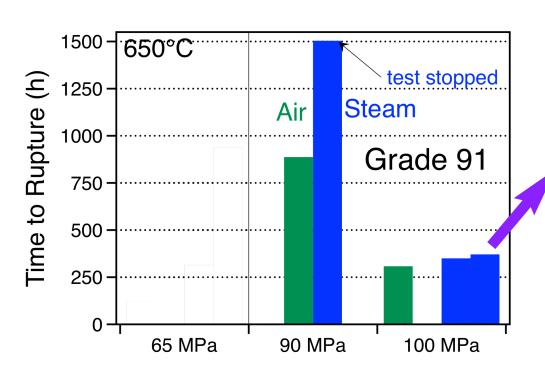


#### two in-situ creep rigs

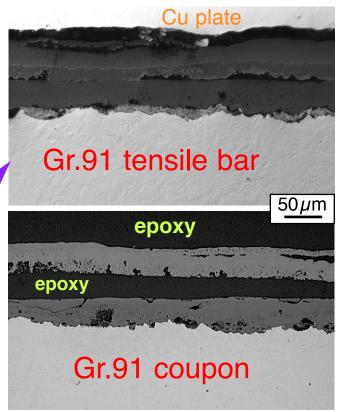


# Grade 91: higher lifetime in steam

650°C 90/100 MPa in air and 1 bar steam



Grade 91: Fe-9Cr-1Mo



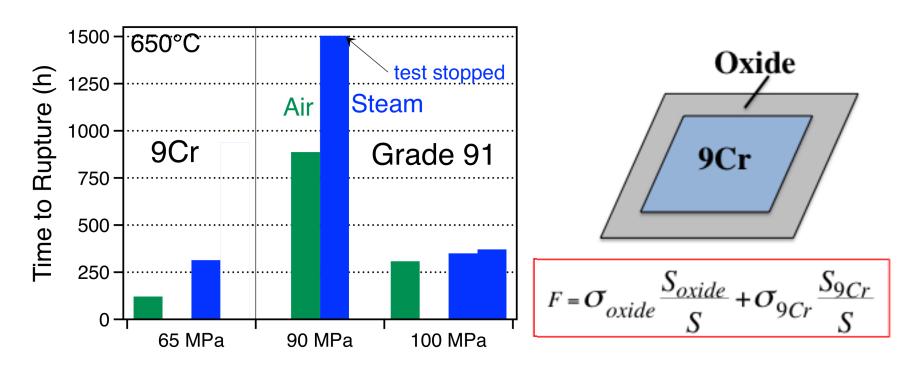
650°C, 500h, 1 bar steam

Verified similar oxide formed on coupon and on creep specimen



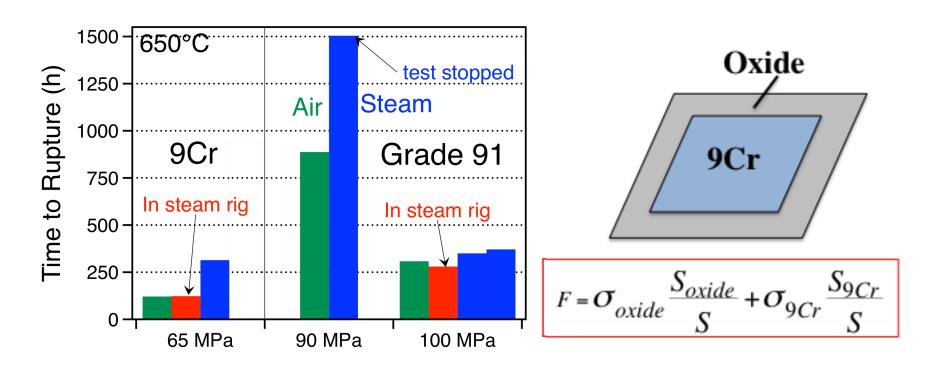
#### More impact for weak substrate

"9Cr" fully ferritic: slow cool from 1100°C



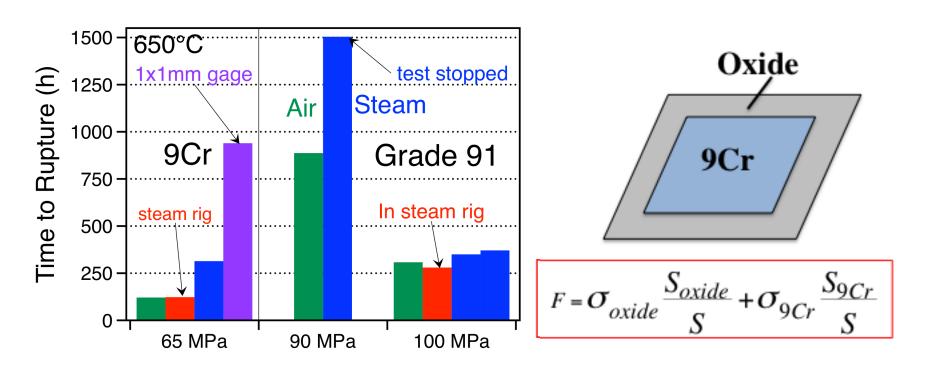
Higher effect of steam when substrate is weaker Suggests that the oxide scale may be load bearing

# Repeated air test in steam rig 650°C 90/100 MPa in air and 1 bar steam



Verified that there was no effect of the in-situ creep rig compared to conventional creep rig test in air

# 8X increase in lifetime for 1mm gage Reduced gage from 2x2mm to 1x1mm



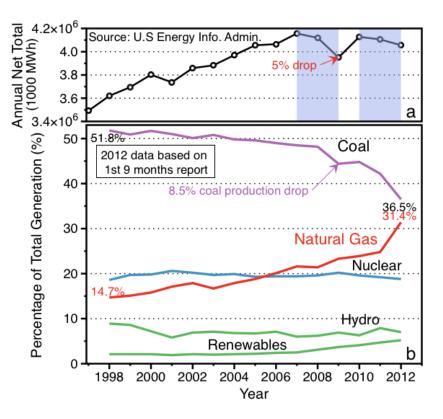
More of the load on the oxide scale with small gage Clear evidence of load-bearing scale on 9Cr steel

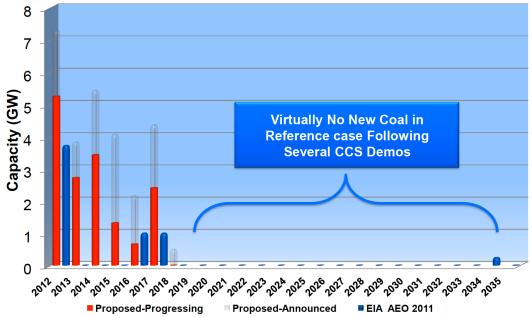
No evidence of creep debit in steam environment Task completed

# Why stop working on oxy-firing?

- no indication of effect
- scarce research \$
- stagnant US demand
- new regulations
- no plants planned

Pint, JOM Aug. 2013





#### FY14: more "real world" issues

- complete oxy-fire reporting
   quantify internal oxidation of NiCr alloys in different environments
- 2) more detailed study of shot peening solution for steam-side scale exfoliation
- 3) H-induced stress corrosion cracking
  - 2.25%Cr steels: Grades 22,23,24
  - significant for industry
  - little "science"
  - need solutions

Cracks in longitudinal direction

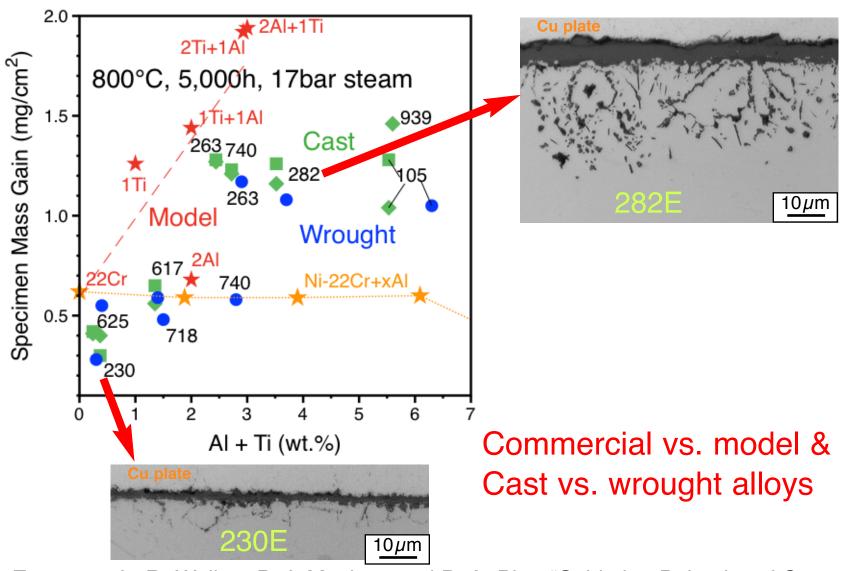


Cracks in transversal direction



## NiCr+Al,Ti: mostly Ti effect

5,000h steam testing at 17 bar, 800°C

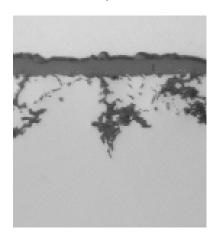


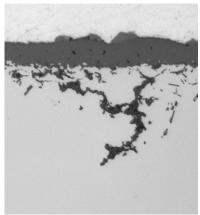
E. Essuman, L. R. Walker, P. J. Maziasz and B. A. Pint, "Oxidation Behavior of Cast Ni-Cr Alloys in Steam at 800°C," Materials Science and Technology, in press.

# Ex: Ni-18Cr-2Al-1Ti comparison

Coupons tested at 800°C in three environments

Steam, 1000h Steam, 2000h Steam, 5000h

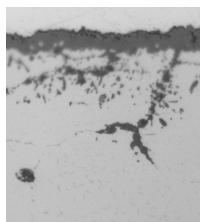




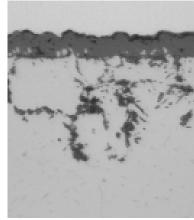


800°C, 5,000h, 17bar steam **282** 105

Wet air, 2000h



Dry air, 5000h



Maximum penetration 617 20 Thickness/Depth (µm)

# Why shot peening?

Exfoliation problem is a main driver for research

H<sub>2</sub>O-accelerated oxidation of steels (steam-side) Tube failures & erosion damage

Cost: planned/unplanned shutdowns, mitigation

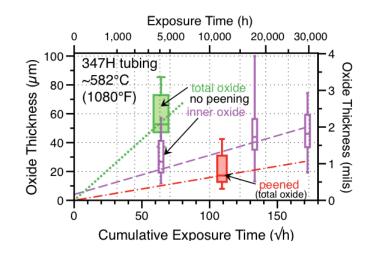
Realization of limited understanding

#### Shot peening of austenitic tubes

Reduced scale growth: avoids exfoliation issue Limited understanding of benefit and procedure

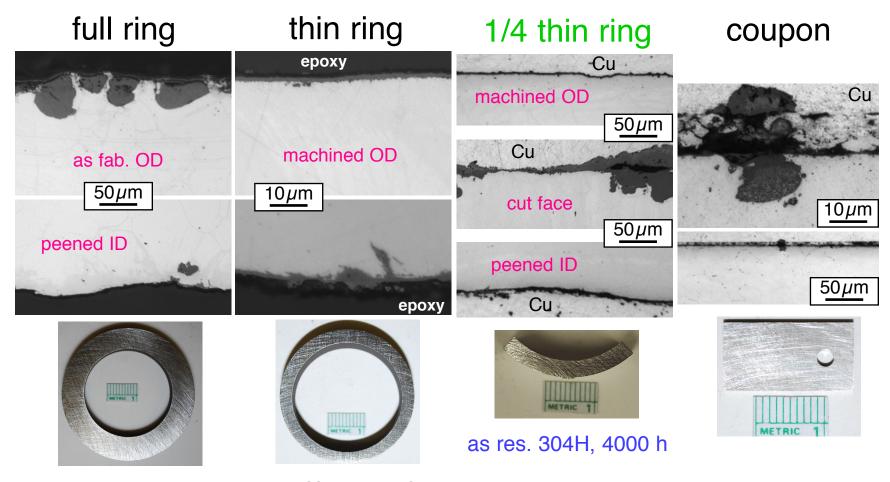
Ex: How do oxide nodules evolve at 600°-650°C?

#### Prior field/lab work with EPRI:





# Specimen type showed minor effects peened 304H 650°C 17bar steam 4,000h

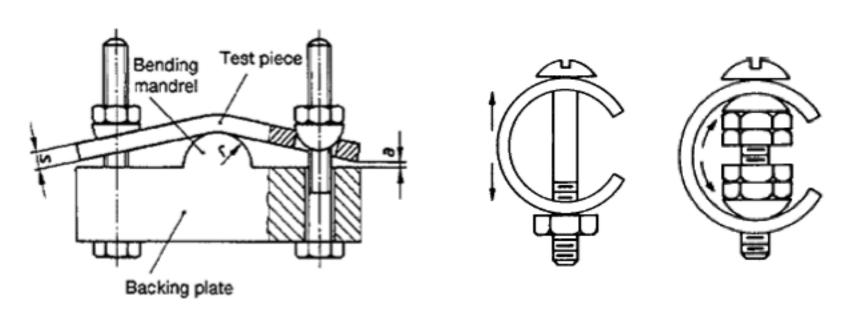


Peened ID: no effect of specimen geometry OD difference: as-received vs. machined (thin ring)

- cold work due to machining similar to peen-

#### Stress corrosion cracking testing

- 1. Stress-environment interaction: 25°-300°C with controlled water chemistry
- 2. Two tests: U.S. (C-ring) and E.U. (Jones) have very different stress states
- 3. Do they produce the same result?



Jones Test

Longitudinal

**C-Ring Test** 

Circumferential

# Both tests showed same results

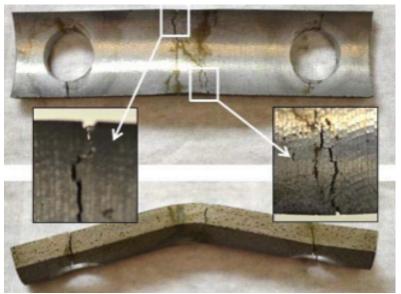
#### 200°C H<sub>2</sub>O±O<sub>2</sub> 168h

	Test Condition				
Alloy	As Received		Normalized		
	Aerated	Deaerated	Aerated	Deaerated	
T23			$\square$		
T24			$\square$		
T92			$\square$		



☑ Cracked

Harder (350 vs 220 DPH), Normalized (0.5h,1065°C WQ) steels are more susceptible to cracking





Normalized T24 after inhibited acid clean

#### FY14+ SCC Testing

- 1. Characterization of cracking
- 2. Do higher strength super-bainitic 3Cr steels show the same behavior?
  - no Grade 315 tubing available
- 3. Are there critical temperature and hardness values for susceptibility?
- 4. Controlled chemistry water loop needed to identify critical O<sub>2</sub> content for cracking
  - loop under construction (due FY15)
- 5. Explore potential solutions

# Milestones FY13

Done - Assess temp. effect on ash testing (12/12)

Done - Complete Ni-Cr+Al, Ti ash testing (3/13)

Done - creep effect of steam-formed oxide (12/13)

Done - Compare SCC C-ring & Jones test (2/14)

#### **FY14**

- Complete Ni-base alloy coal ash eval. (2014)
- Assess shot peening at 600°-650°C (2014)
- Compare SCC of 2.25Cr and 3Cr steels (2014)

# Summary

ORNL corrosion task transitioning Prior focus on oxy-firing + in-situ creep testing Wrapping up: oxy-firing no worse than air Model NiCr alloys: Cr, Co, Mo, Al, Ti effects No detrimental effect of steam on creep FY13-14 transitioning to "real world" issues Shot peening - study exfoliation solution SCC - study current waterwall issue Followup - quantify internal oxidation of NiCr - compare air, wet air, steam

# GLEAN GOAL. GOOL.



# Ash experiment issues

#### **Experiments:**

- air/oxy: worst case comparison
- "milder" oxy-firing: lower H<sub>2</sub>O or SO<sub>2</sub>
- add cold recirculation: low H<sub>2</sub>O/low SO<sub>2</sub>

#### Test protocols to be evaluated:

- use of Pt catalyst
- crucible (covered sample) vs. ash slurry
- cycle frequency 10 x 100h vs. 500h x ?
- goal: "actual" rate or accelerated?

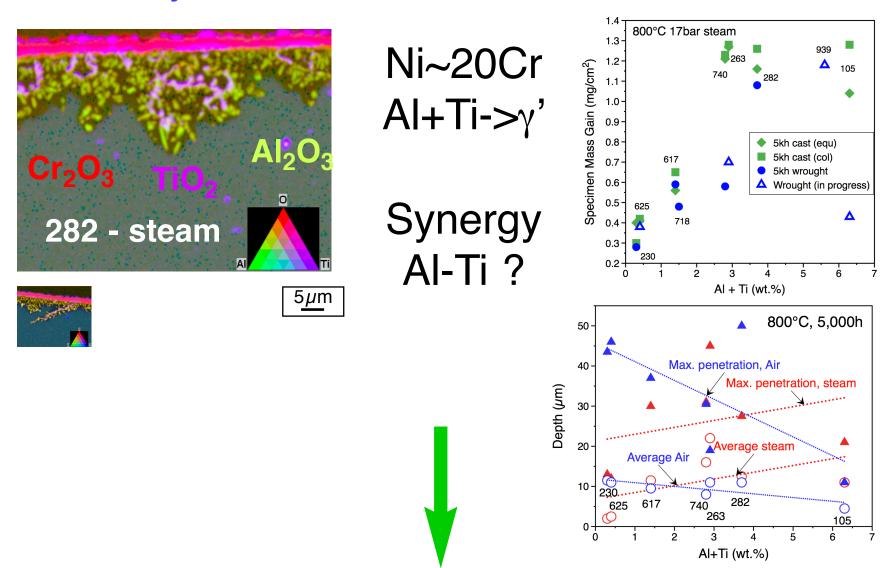
#### Metal loss measurement

- box plots capture variable attack
- scale thickness (not rod diameter)

Ash composition: how changed by oxy-firing?

# 800°C steam follow up work

Alloy 282: 5kh in 17bar steam or lab. air



Model alloys: Ni-22Cr + Al +/- Ti in steam

#### What's different here?

Many previous & current studies of oxy-firing & CO<sub>2</sub>

- "Oxy" worse: Speigel (2006) + Corvino (2008)
- Complicated: boiler OEMs have advantage
- CO<sub>2</sub> effect: Jülich, U. Pitt & Australia (Young)

#### Issues with fireside corrosion experiments:

Different experimental conditions (if published)

1000h vs. 10 x 100h (ash re-supply)

Ash/gas/temp. variables

Use of Pt catalyst (SO<sub>2</sub>/SO<sub>3</sub>)

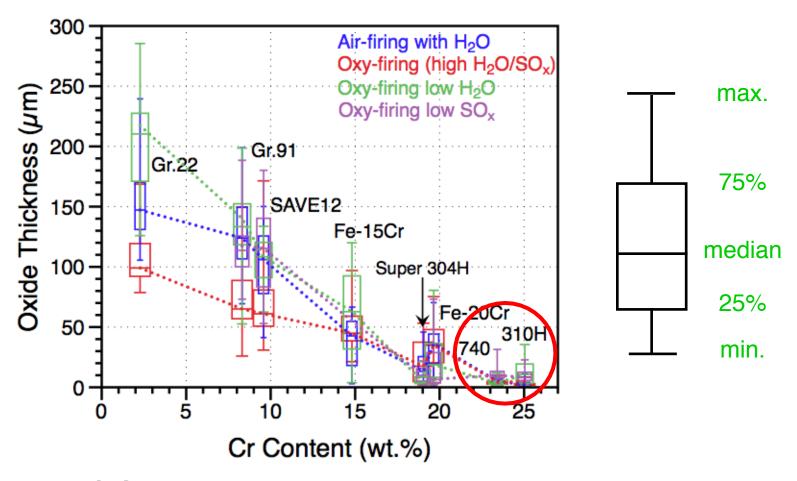
\* Evaluate experimental parameters (FY12)

Typically, only commercial alloys evaluated
Prior work showing Cu-containing alloy attacked
Was it an effect of Cu or other element(s)?

\* Model alloys to better understand composition

# Little effect of gas at 600°C

Synthetic coal ash, 500h exposures in 4 gases

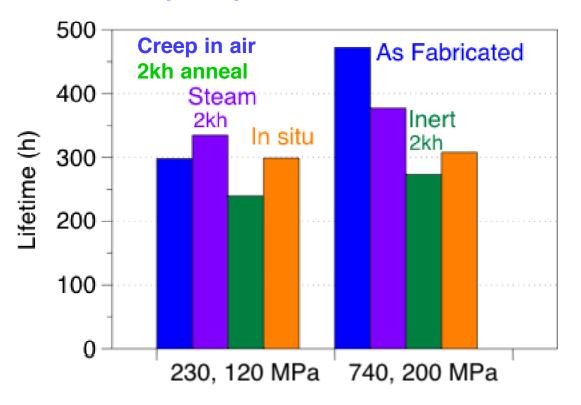


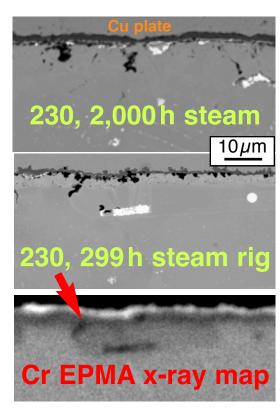
Higher CO<sub>2</sub> environments not detrimental Expected the lower SO<sub>2</sub> environment to lower attack

- same synthetic ash used in all cases

#### 800°C: 230/740 limited steam effect

Creep rupture tests in air and 1 bar steam





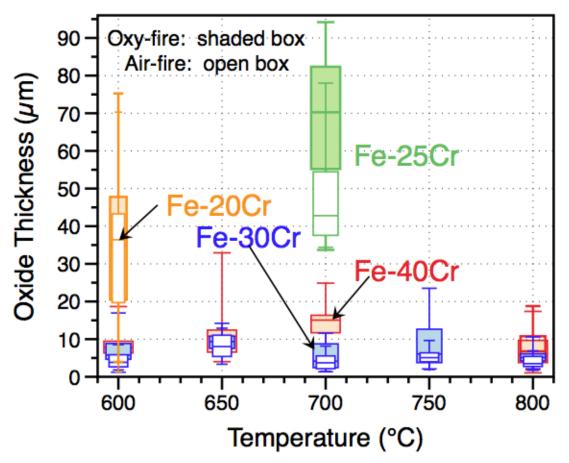
230: no effect of in-situ or ex-situ steam

740: microstructural reason for decreased life?

- alloy/oxide characterization in progress
- task will conclude this summer with paper

#### Model Fe-Cr alloys show Cr benefit

Provide guidance for Cr-rich coatings

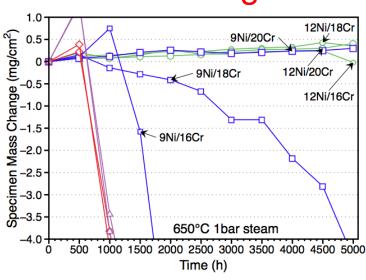


Fe-30Cr chosen for more study Fe-40Cr casting had problems with Cr rich phases Currently, filling in matrix for Fe-20Cr and 25Cr.

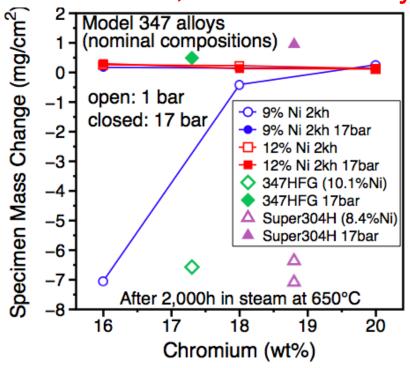
# Model 347 alloys: 650°C steam

Cast, hot rolled Fe-Cr-Ni-1.5Mn-0.4Si-0.8Nb-0.09C

#### 1 bar mass gain:



#### 1 & 17bar 2,000h summary:

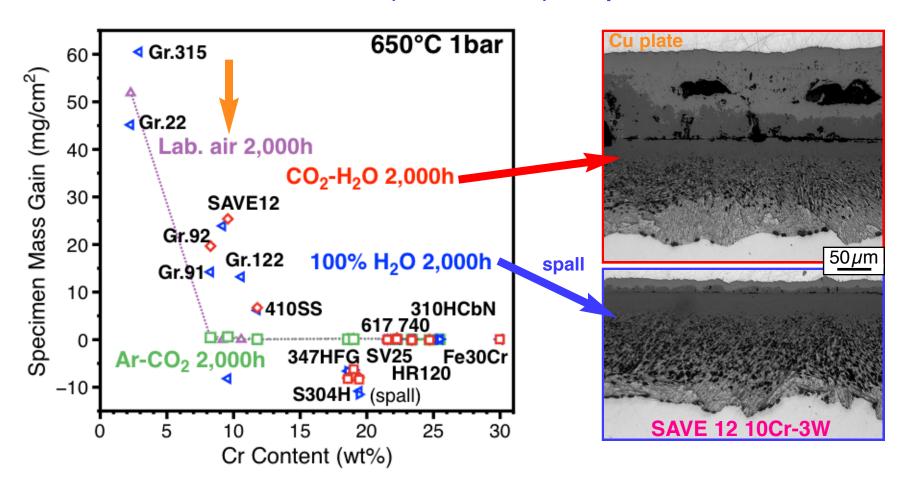


5,000h 1bar exposure completed in March 2012 Higher (12%) Ni content very beneficial 2000h 17bar completed April 2012 (no effect)

Concern: model alloys better than 347HFG & S304H

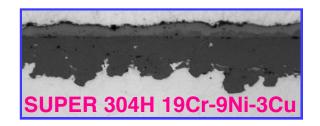
### 650°C: CO<sub>2</sub>-H<sub>2</sub>O not any worse

After 2,000h (4 x 500h) exposures



Spalled outer oxide in 100%H<sub>2</sub>O (both SAVE 12 & Super304H)

Pint & Thompson, Mater. Corr. in press



#### NiCr+Al, Ti 800°C: complete destruction

Coupons: 500h in synthetic coal ash + "oxy-firing" gas

