

# Development of Alumina-Forming Austenitic (AFA) Stainless Steels

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*“23<sup>rd</sup> Annual Conference on Fossil Energy Materials,” at Pittsburgh, PA  
May 12-14, 2009*

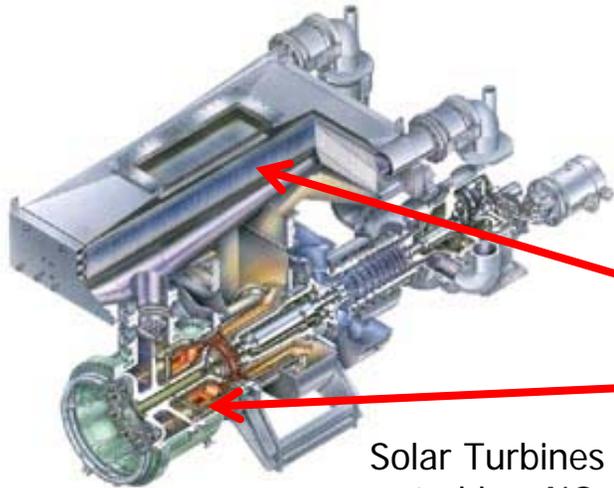
# Stainless Steels with Higher-Temperature Capability Needed

- Driver: Increased efficiencies with higher operating temperatures in power generation systems.
- Key issues are **creep** and **oxidation resistance**.
  - Significant gains have been made in recent years for improved creep resistance via nano MX precipitate control (M = Nb, Ti, V; X = C, N).
  - Stainless steels rely on  $\text{Cr}_2\text{O}_3$  scales for protection from high-temperature oxidation.
    - Limited in many industrial environments (water vapor, C, S)
    - Most frequent solution is coating: costly, not always feasible

# Development Effort for Low Cost, Creep and Oxidation-Resistant Structural Alloy for Use from ~600-900°C

- **Approach:  $\text{Al}_2\text{O}_3$ -forming austenitic stainless steels**  
-background and potential advantages
- **Current alloy status for microstructure, mechanical properties, and oxidation resistance**

## Initial target(s)



Solar Turbines 4.6 MW Mercury 50 recuperated low NO<sub>x</sub> gas turbine engine

**Fossil Power  
Steam Turbine,  
Boiler Tubing**

**Recuperator,  
Casing**

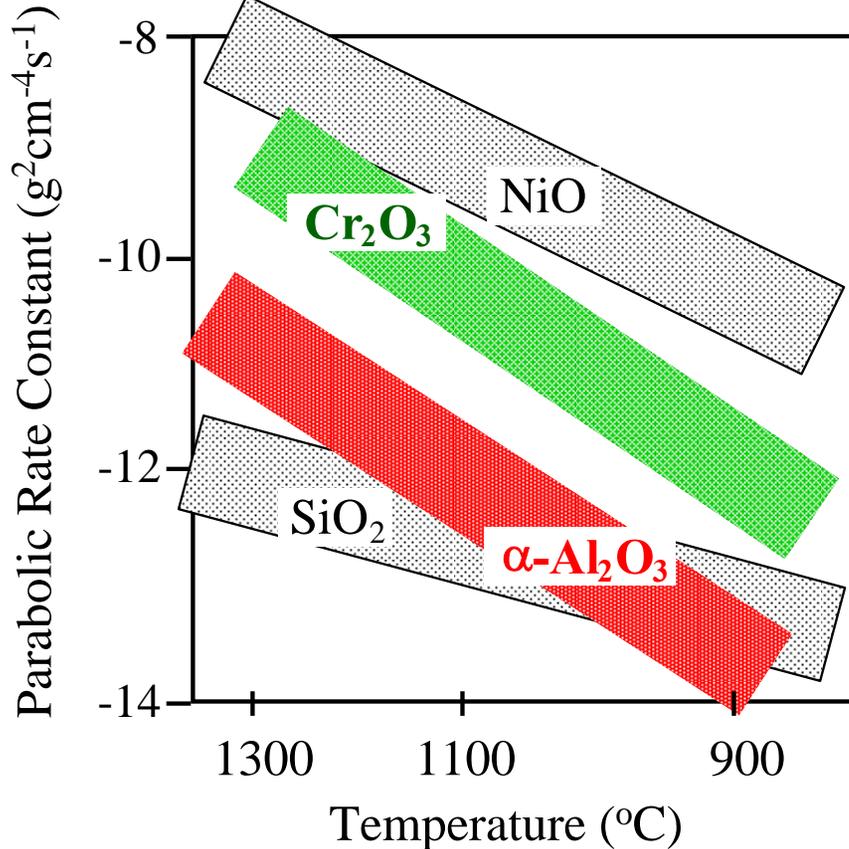


HMN-Series (High-, Intermediate- and Low-Pressure) Steam Turbine for Combined-Cycle and Steam Power Plants

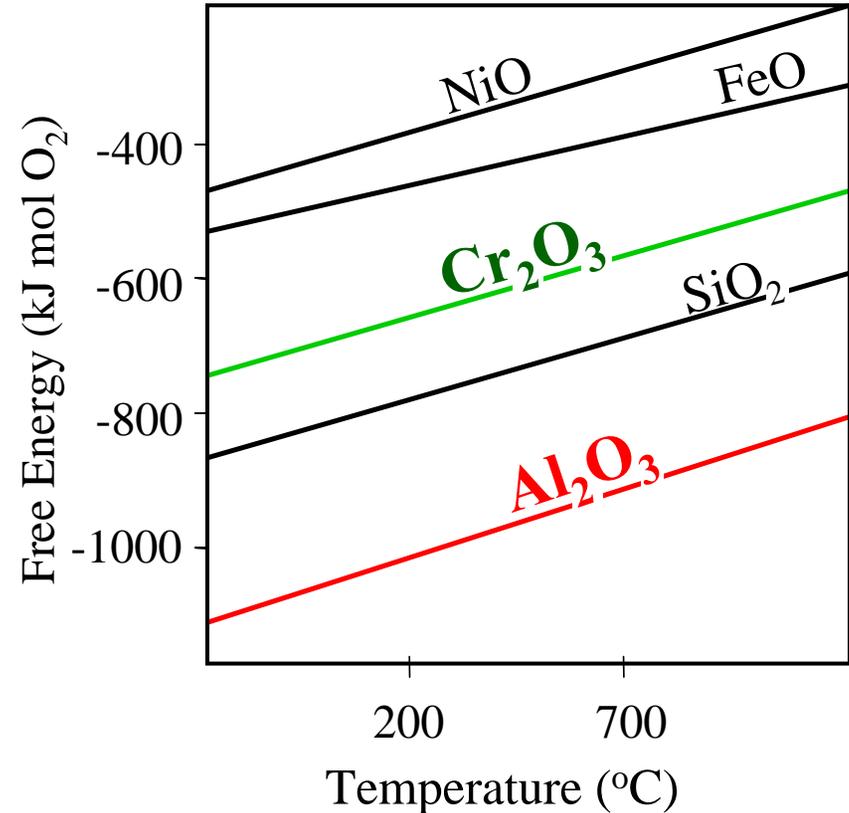
➤ Tubing in chemical/process industry, etc. also targeted.

# $\text{Al}_2\text{O}_3$ Scales Offer Superior Protection in Many Industrially-Relevant Environments

=Kinetics=  
(Growth rate of oxide scales)



=Thermodynamics=  
(Ellingham diagram)



- $\text{Al}_2\text{O}_3$  exhibits a lower growth rate and is more thermodynamically stable in oxygen than  $\text{Cr}_2\text{O}_3$ .
- Highly stable in water vapor.

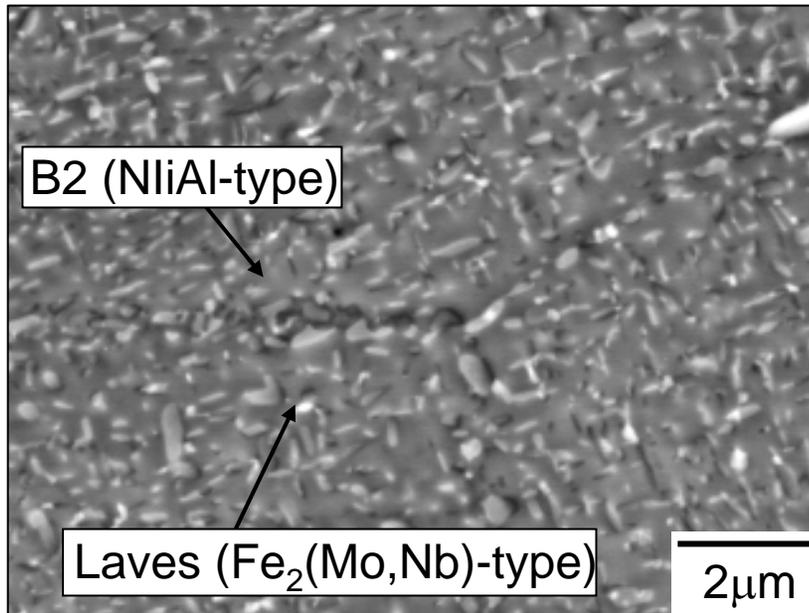
# Challenge of Alumina-forming Austenitic (AFA) Stainless Steel Alloys

- Numerous attempts over the past ~30 years (e.g. McGurty et al. alloys from the 1970-80's, also Japanese, European, and Russian efforts)
- Problem: Al additions are a major complication for strengthening
  - strong BCC stabilizer/delta-ferrite formation (weak)
  - interferes with N additions for strengthening
- Want to use as little Al as possible to gain oxidation benefit
  - keep austenitic matrix for high-temperature strength
  - introduce second-phase (intermetallics/carbides) for precipitate strengthening

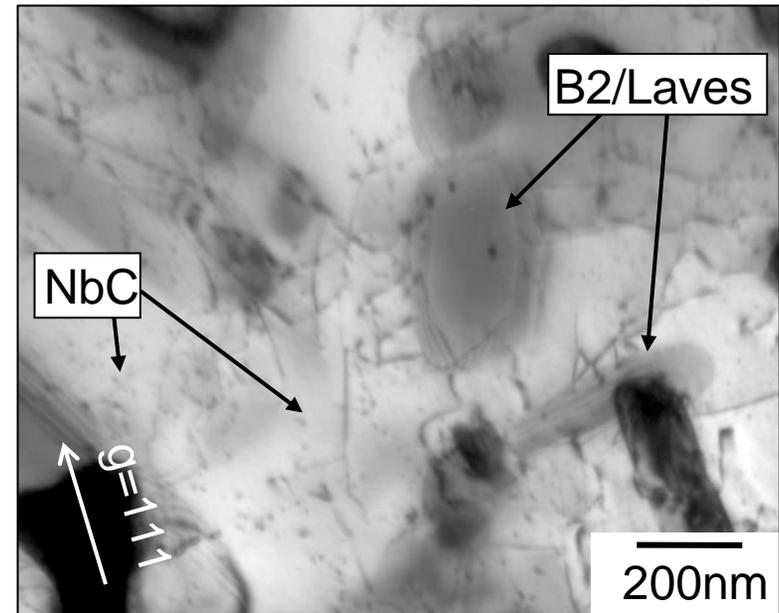
# Composition and Microstructure Considerations for AFA Stainless Steels

Typical Fe-(20-30)Ni-(12-15)Cr-(2.5-4)Al-(1-3)Nb-0.1C wt.%  
Base AFA Alloy Microstructure After Creep

SEM



TEM



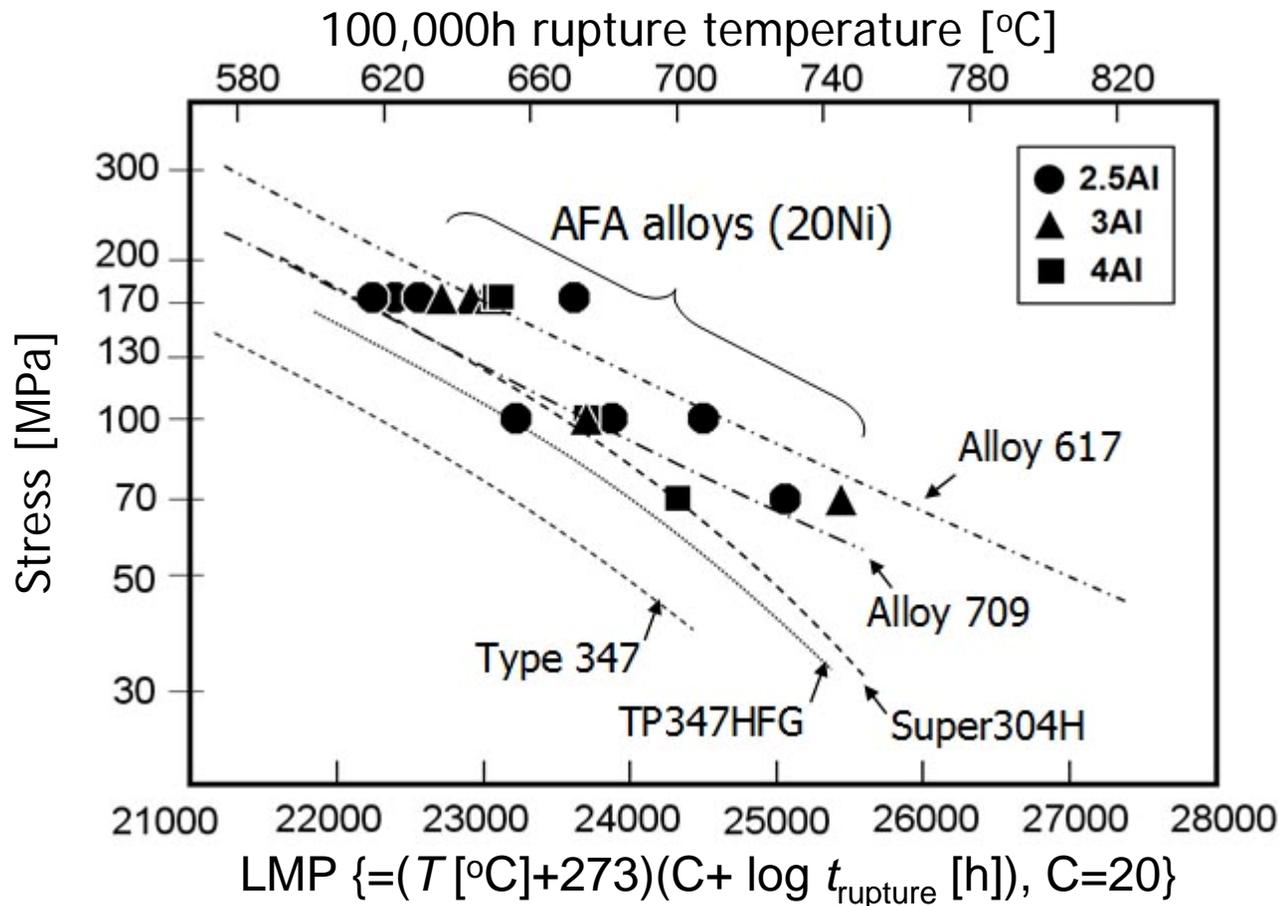
## • Creep Strength

- balance Al, Cr, Ni, to maintain single-phase FCC austenitic matrix
- Nano NbC and submicron B2-NiAl +  $\text{Fe}_2\text{Nb}$  base Laves precipitates

## • To form protective alumina:

- Ti+V < 0.3 wt.%; Nb > (0.6-1) wt.%; N < 0.02 wt.%

# AFA Exhibits Comparable Creep Strength to Best Commercial Austenitic Stainless Steels



- AFA alloys (20Ni) are in the range between alloy 709 (Fe-20Cr-25Ni base) and alloy 617 (Ni-22Cr-12Co-9Mo base)
- AFA data from small (< 1 kg) laboratory arc-castings, sub-sized screening creep test sample, solution treated + 10% cold work condition

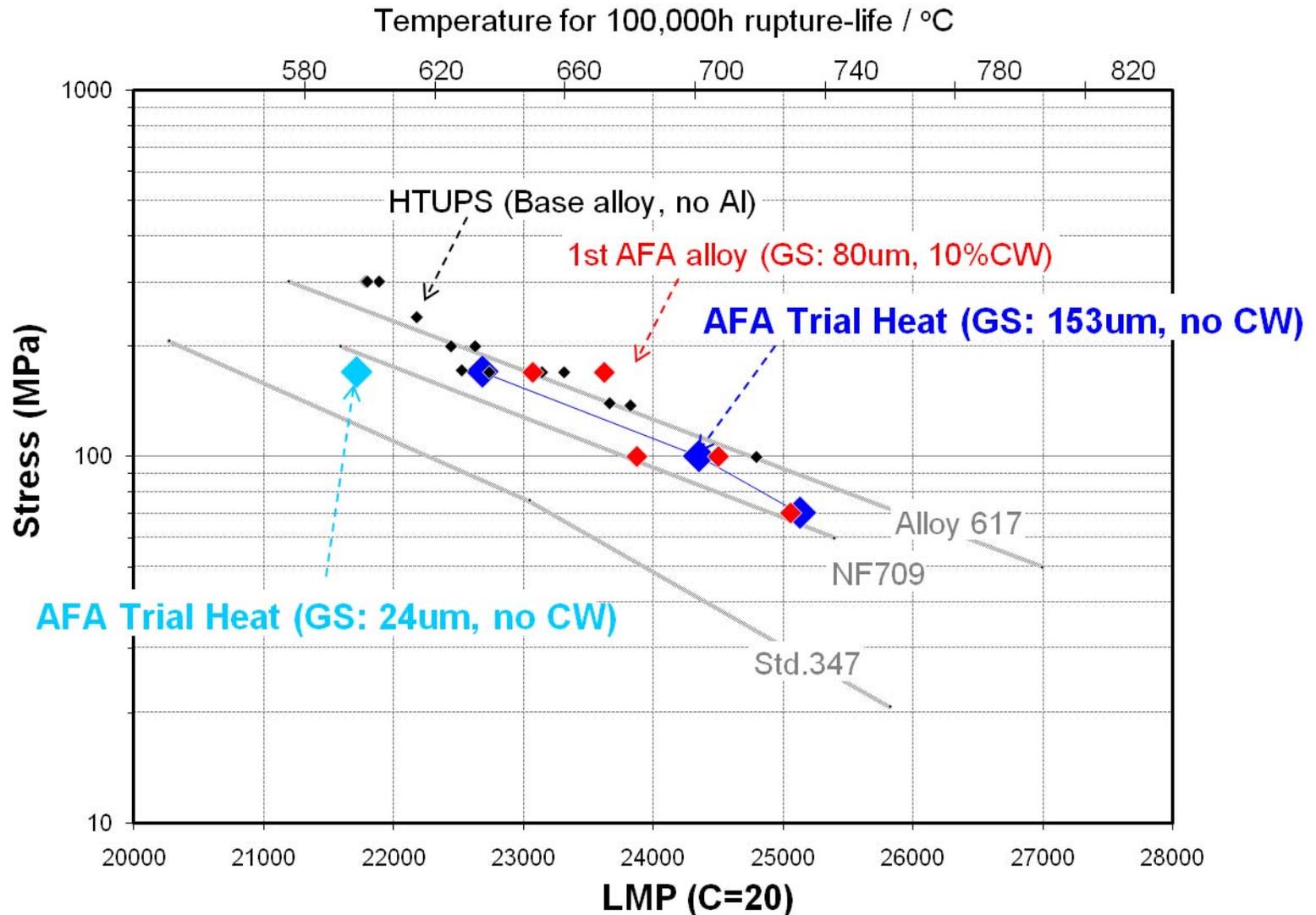
# 50 lb AFA Trial Heat Made by Conventional Vacuum Melting and Hot Rolling Processes



- Fe-20Ni-12Cr-4Al-0.6Nb-0.1C base wt.% composition
- Material used for rigorous creep evaluation with standard specimen design, solution treated condition (no cold work)  
(collaboration w/J.P. Shingledecker)

# Comparable Creep Strength to that Obtained in Screening Studies

(Hot-rolled + Solution heat-treated, no cold-work applied)



# Ductile Creep Rupture at Failure

(Most of the gage portions showed just tinted color even after >6000h testing)

(750°C/170MPa)  
**150h /55%**



\* Almost no damage on the surface, because of relatively shorter testing time.

(750°C/100MPa)  
**6382h/7.5%**  
%



\* Heavily oxidized only at necking regions.

(800°C/70MPa)  
**2634h/28%**



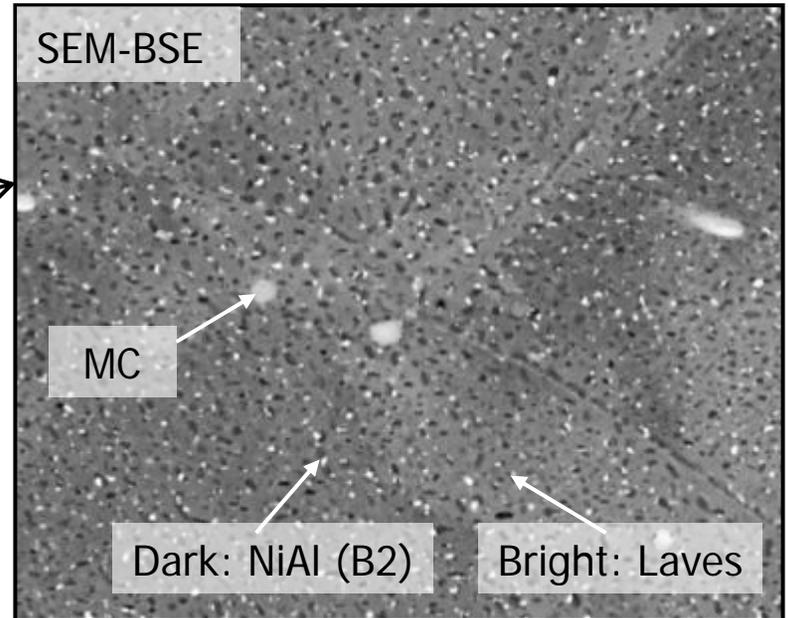
20mm

# Intermetallics Appeared in Early Stage of Creep-testing

(750°C/170MPa) **150h /55%**



5mm



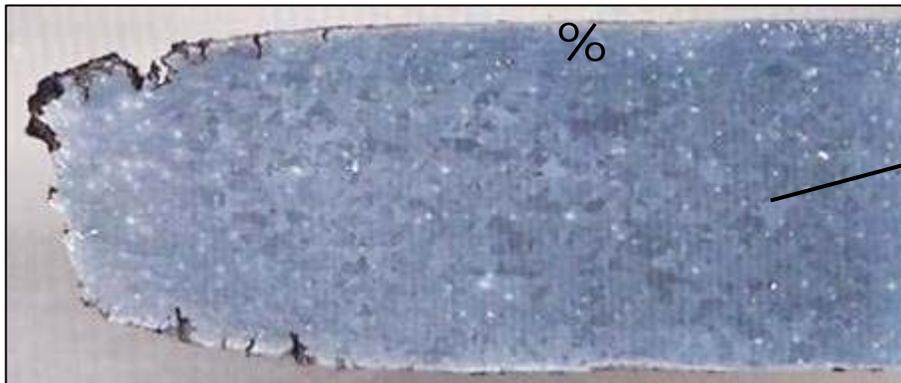
SEM-BSE

MC

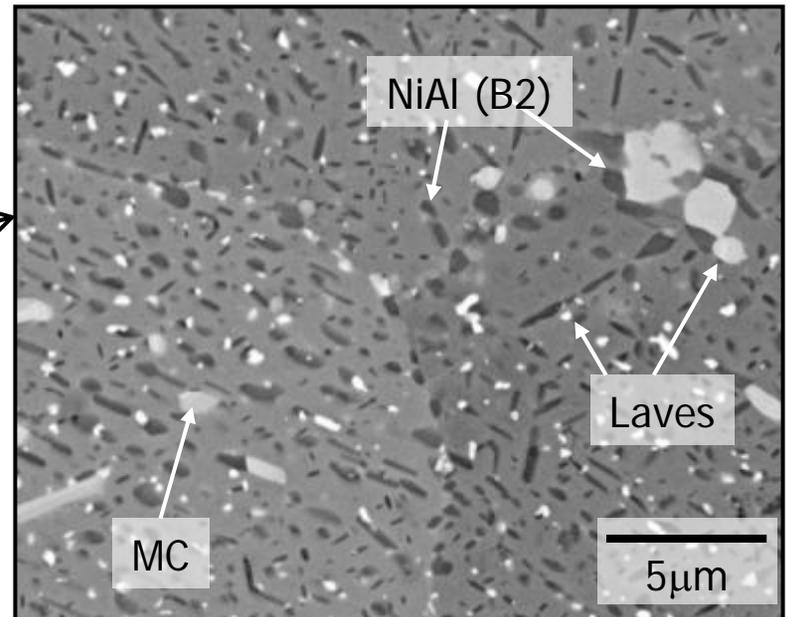
Dark: NiAl (B2)

Bright: Laves

(750°C/100MPa) **6382h/7.5**



5mm



NiAl (B2)

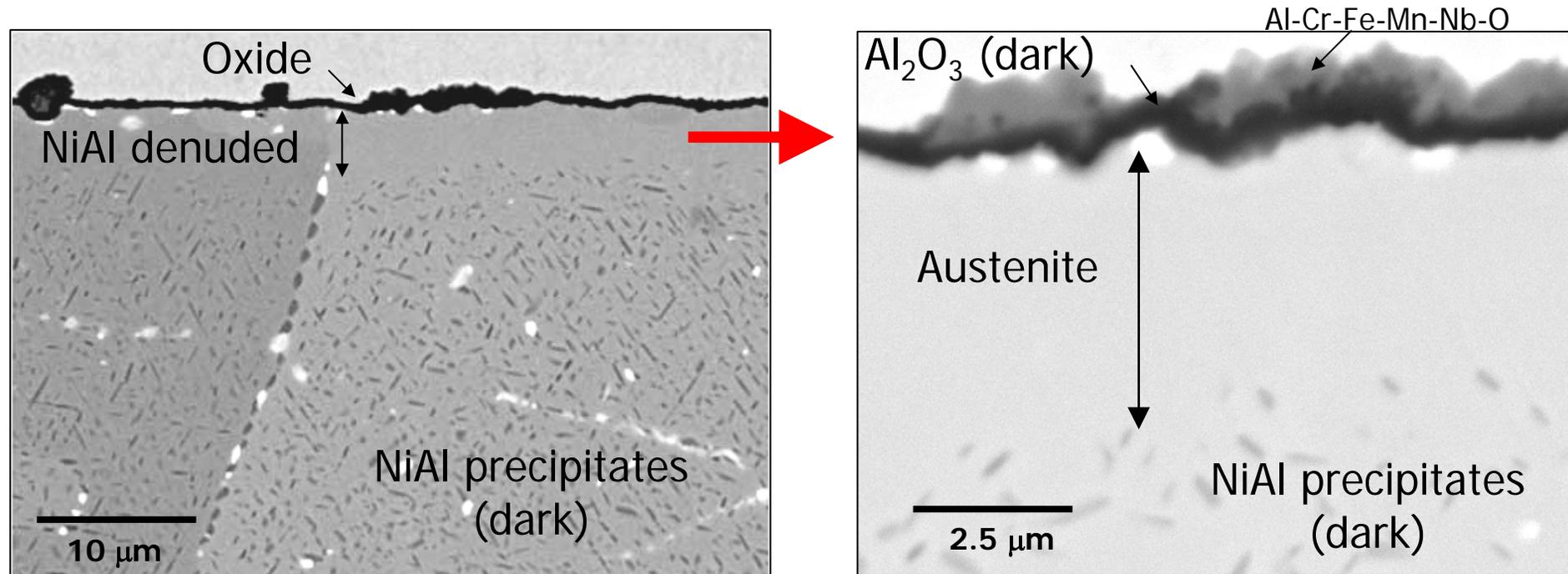
MC

Laves

5µm

# Austenite Matrix and NiAl Precipitates Key to Establishing and Maintaining Alumina

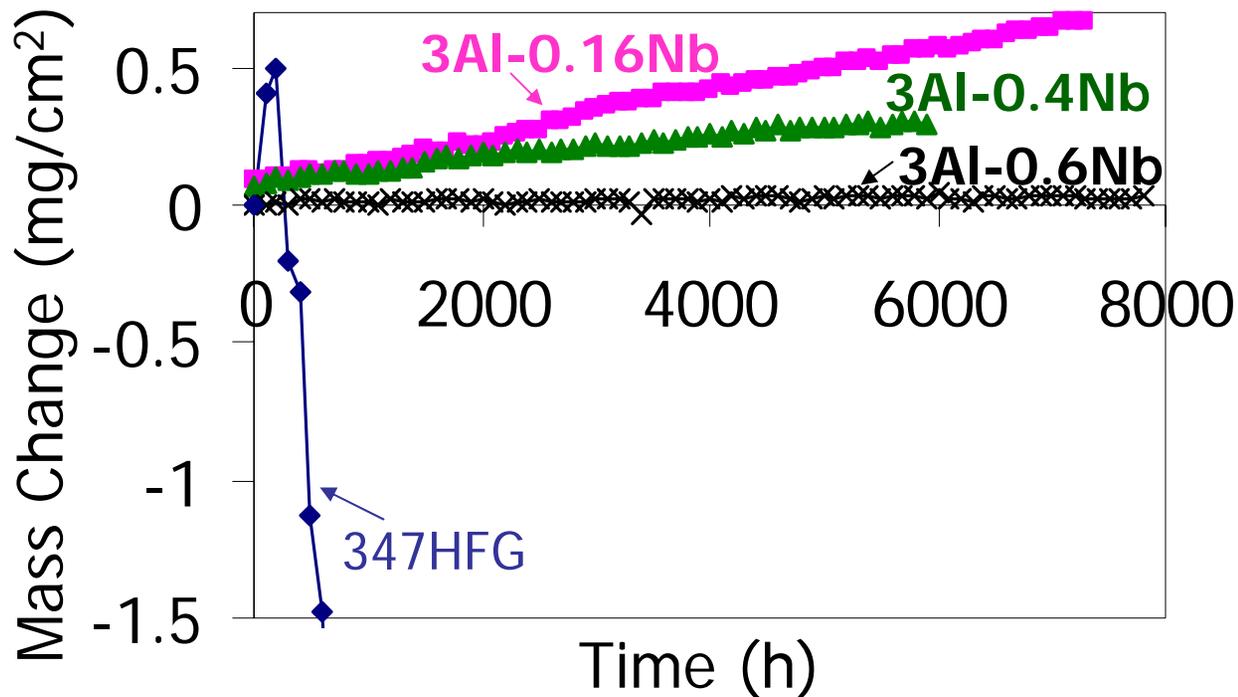
SEM-BSE Images of Typical Oxidized Cross-Section for a 4 Al wt.% AFA Alloy (900°C/100h/in air)



- Austenite matrix composition key to forming alumina
- NiAl precipitates act as Al reservoir to maintain alumina

# Higher Nb in Alloy Favors Better Oxidation Resistance in Air + Water Vapor

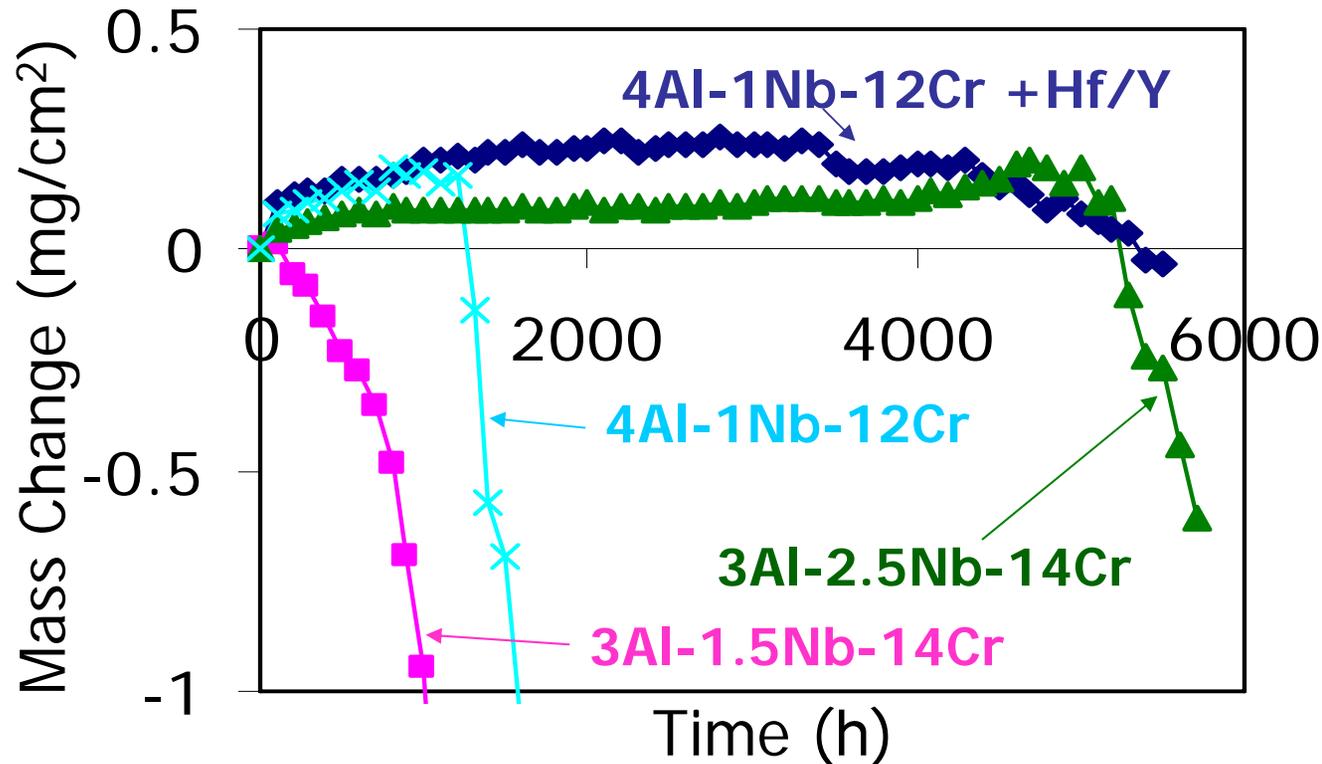
Oxidation at 650°C in Air + 10% Water Vapor



- Excellent resistance out to ~8000 h of ongoing exposure
  - 347 stainless steel shows accelerated attack after a few hundred hours under these conditions

# Increased Nb or Hf/Y Additions Aide $\text{Al}_2\text{O}_3$ Formation at 800°C in Air + Water Vapor

## Oxidation at 800°C in Air + 10% Water Vapor

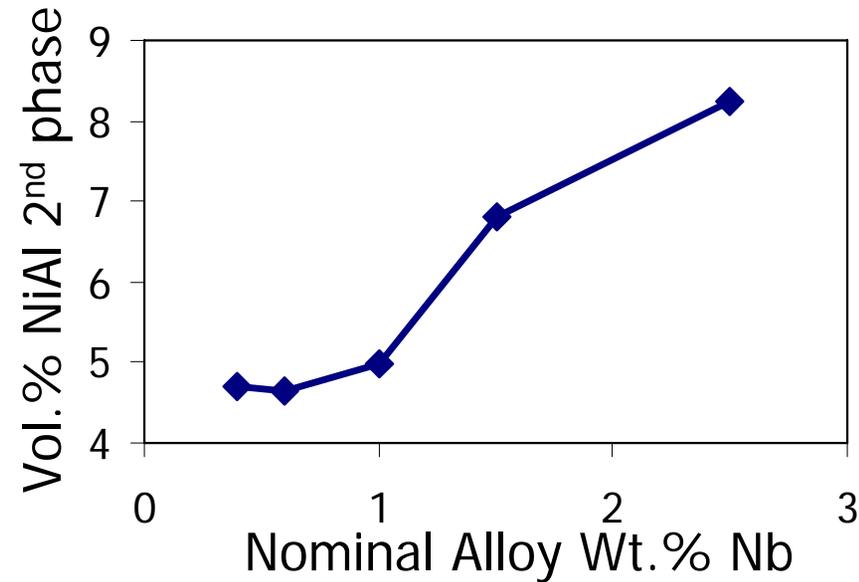
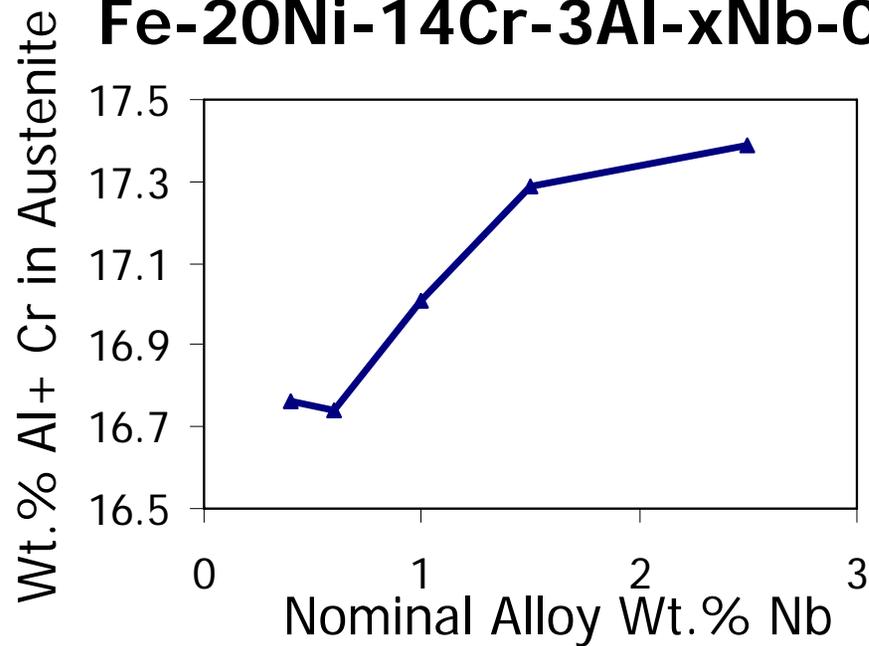


• Best alloys still showed transition to Fe oxide nodule formation and mass loss

-upper temperature limit  $700^\circ\text{C} < t < 800^\circ\text{C}$  in  $\text{H}_2\text{O}$  for these alloys

# Hypothesized Microstructural Benefits of Nb for Improved Oxidation Resistance

## Computational Thermodynamic Predictions for Fe-20Ni-14Cr-3Al-xNb-0.1C wt.% Base (750°C data)

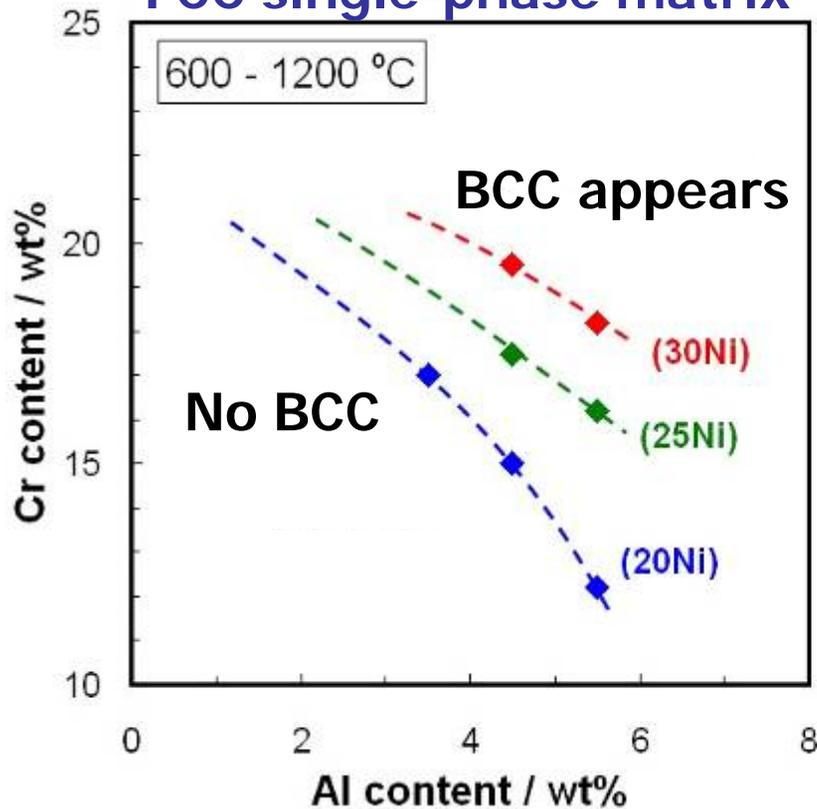


- Nb increases relative Cr + Al levels in austenitic matrix to help form  $\text{Al}_2\text{O}_3$ 
  - Cr aides formation of alumina via third-element effect
- Nb increases B2-NiAl precipitate volume fraction (Al reservoir for  $\text{Al}_2\text{O}_3$  scale)
- Insights used to design next iteration of AFA for higher service temperature

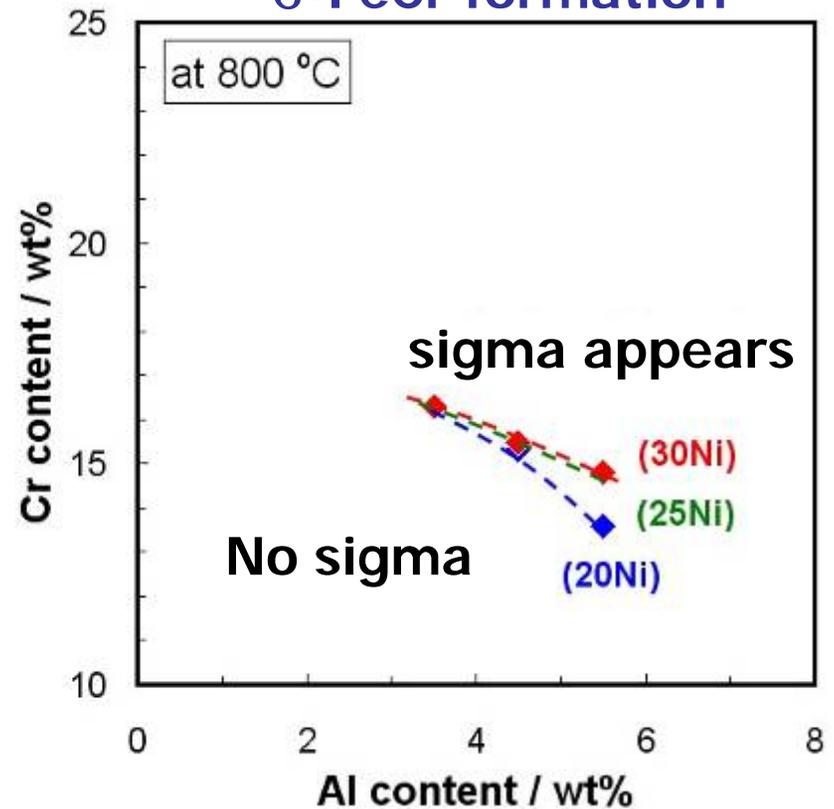
# Thermodynamic Calculations Predict Al-Cr-Ni Balance for Austenite and $\sigma$

Fe-Cr-Ni-Al-1Nb-2Mo-0.1C (wt%)

FCC single-phase matrix



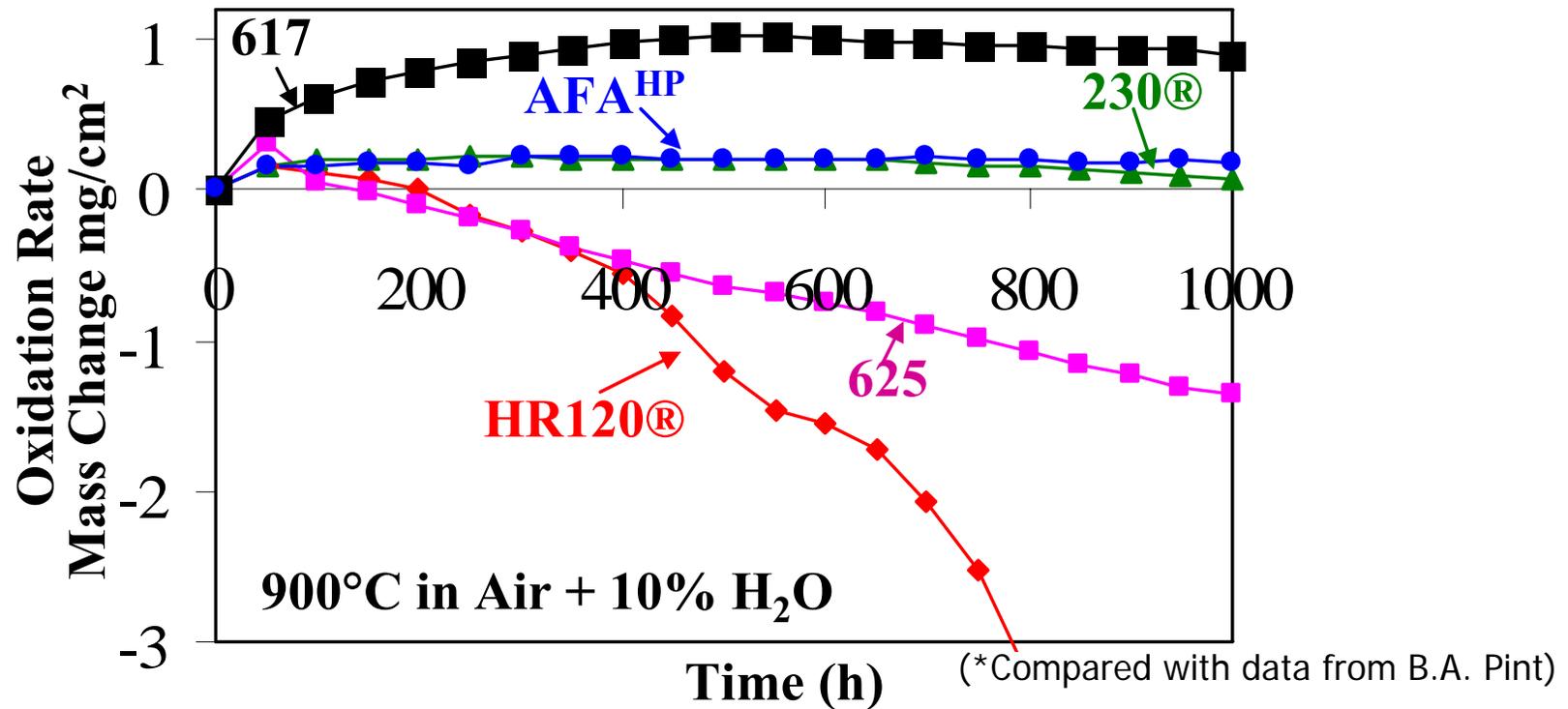
$\sigma$ -FeCr formation



- Higher Cr levels possible at 4 Al and 25 Ni than early 4Al-12Cr alloy series
- Increased sigma risk w/decreasing temperature below 800°C
  - counter balance with lower Mo, Nb, and W

# Recent High Al and Cr AFA<sup>HP</sup> Alloys Show Promise to 900°C in Air + 10% H<sub>2</sub>O

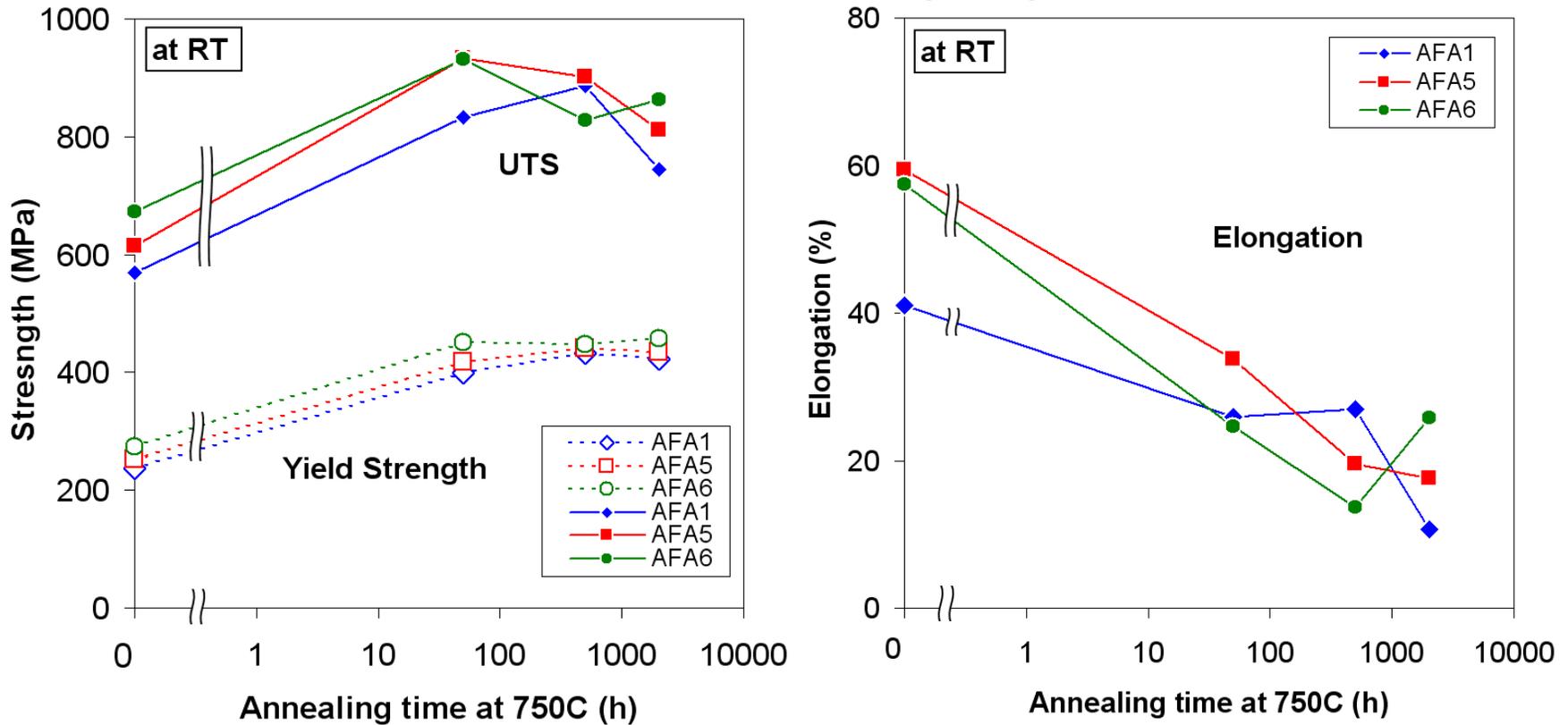
Cyclic Oxidation (10 h cycles) at 900°C in air + 10% H<sub>2</sub>O



- AFA<sup>HP</sup>: Fe-25Ni-(14-15)Cr-4Al-2.5Nb-Hf/Y wt.% base
  - Good 900°C behavior also observed at 3-3.5Al wt.% and no Hf/Y addition
- Better resistance than more expensive HR120 (~Fe-35Ni-25Cr) and Ni-base alloys 625 and 617 under this test condition

# Room-Temperature Tensile Evaluation as a Function of 750°C Ageing Time

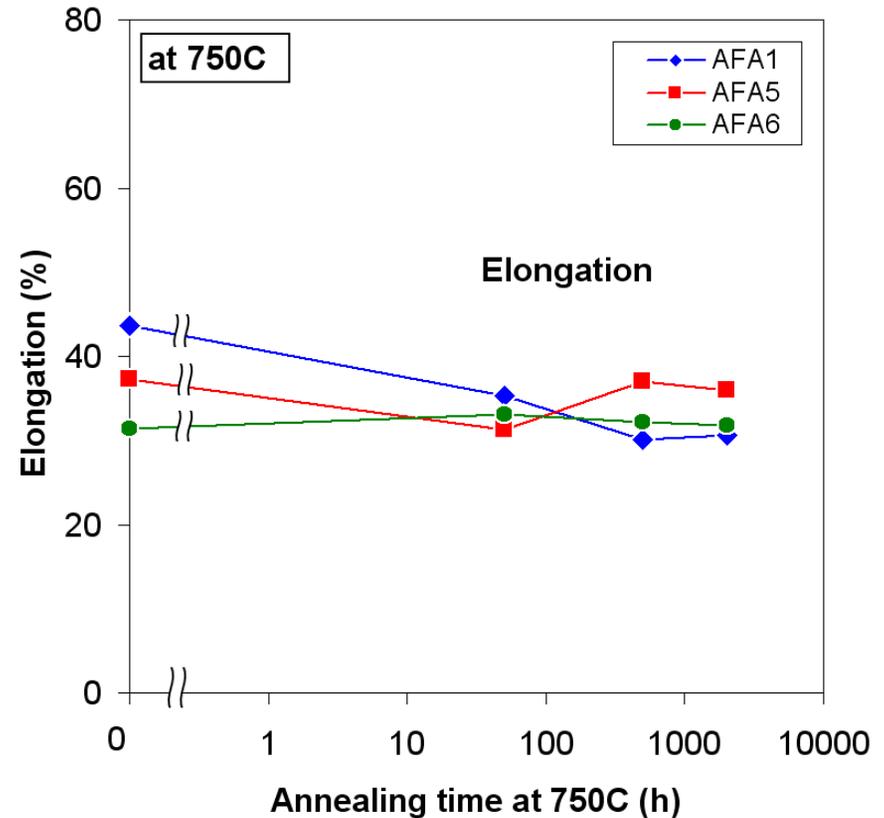
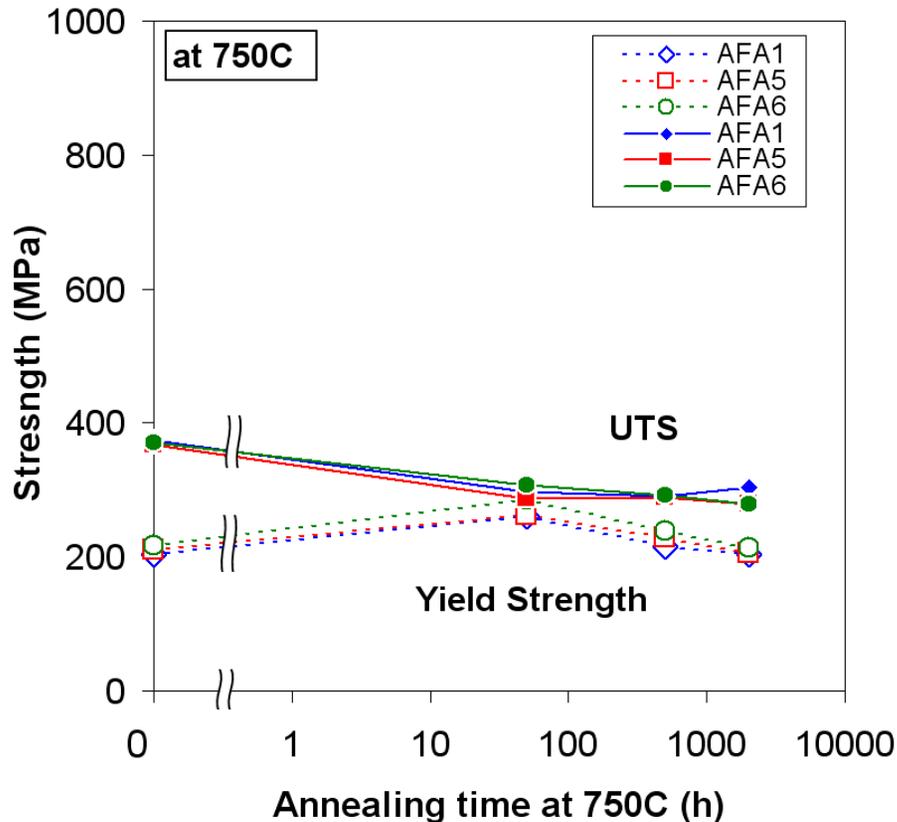
Fe-(20-25)Ni-(3-4)Al-1Nb base wt.% Alloys Evaluated at Room Temperature After Ageing at 750°C



**At room temperature:**

- Yield strengths reach maximum after ageing ~ 50 hour at 750°C
- Elongation to fracture decreases with ageing: ~10-20% elongation retained

# Little Effect of 750°C Ageing on Tensile Behavior at 750°C



- Lower yield/ultimate strength at 750°C than at room temperature
- Elongation unaffected by ageing

# Summary

- A new class of Fe(Ni)-base,  $\text{Al}_2\text{O}_3$ -forming, high creep strength austenitic stainless steel alloys is under development
- Excellent oxidation resistance observed in air +  $\text{H}_2\text{O}$ 
  - All AFA alloys have upper-temperature limit for  $\text{Al}_2\text{O}_3$  formation (consequence of low Al + Cr to also achieve mechanical properties)
  - ~650-900°C in air +  $\text{H}_2\text{O}$  depending on alloy composition
- Promising mechanical properties
  - Creep resistance comparable to best available austenitics
  - High tensile elongation in solution treated condition
  - 10-20% ambient tensile elongation retained on ageing
  - trial heats show good properties comparable to lab scale castings

# Future Work

- Spin-off demonstration project under EERE for AFA foil in turbine recuperator applications (3 CRADAs signed)
- Continued AFA development under Fossil (funding permitting)
  - linked experimental and modeling efforts directed toward improved understanding of AFA microstructure, oxidation, and creep to provide basis for further alloy development
  - long-term studies of creep and oxidation to provide basis for transition to industry
  - expand and evolve AFA concept towards development of alumina-forming, Fe-base superalloys

## Acknowledgments

The Office of Fossil Energy, U.S. Department of Energy, National Energy Technology Laboratory, under Contract DE-AC05-00OR22725 with UT-Battelle, LLC, and The SHaRE User Facility in Oak Ridge National Laboratory, which is sponsored by the Division of Scientific User Facilities, Office of Basic Energy Sciences, U.S. Department of Energy.

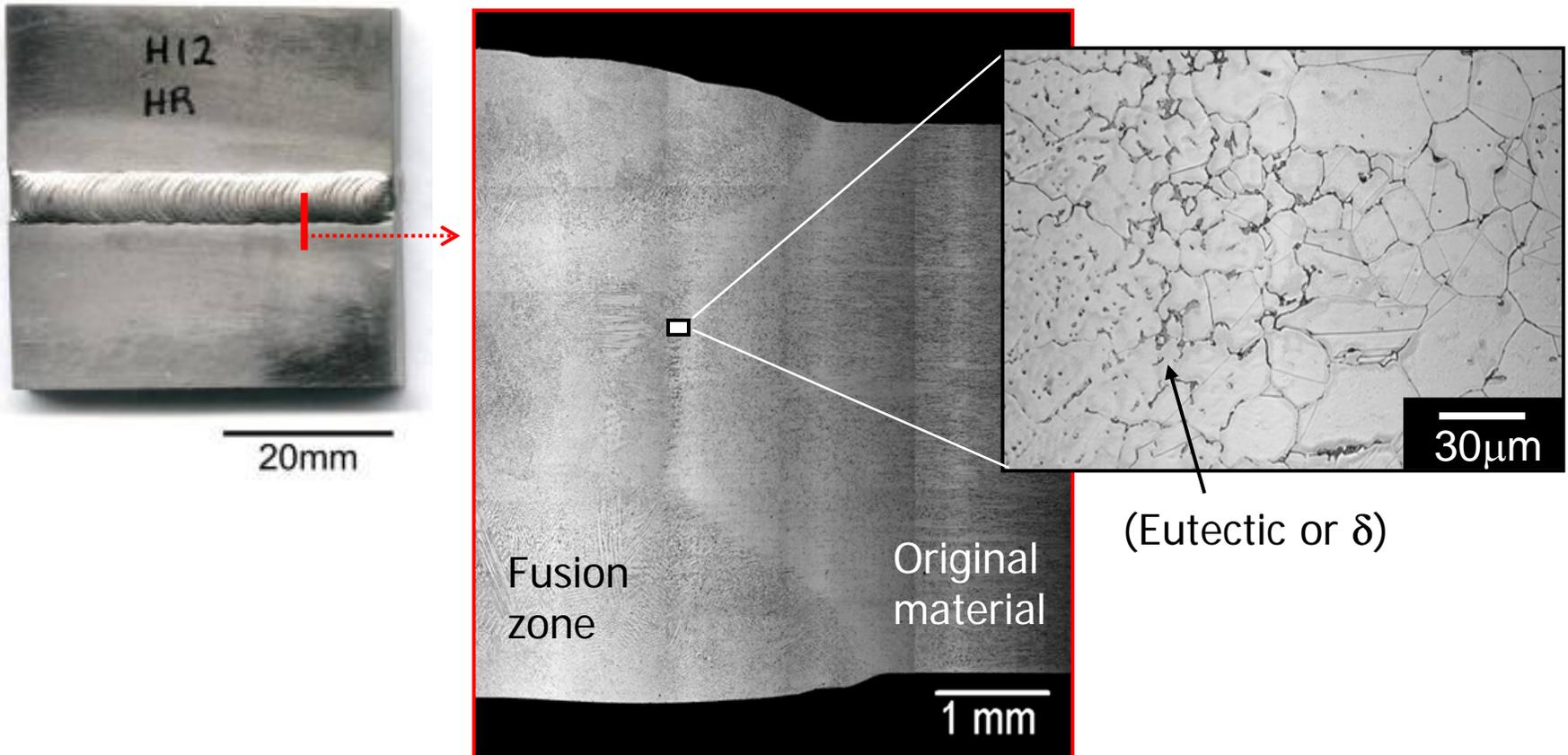
# Three Grades of AFA Alloys Identified Thus Far

- **AFA Grade: Fe-(20-25)Ni-(14-15)Cr-(2.5-3.5)Al-(1-3)Nb wt.% base**
  - ~750-800°C temperature limit for Al<sub>2</sub>O<sub>3</sub> formation
  - Trial heats with commercial alloy producer
- **High Performance AFA<sup>HP</sup> Grade: Fe-(25-30)Ni-(14-15)Cr-(3.5-4.5)Al-(1-3)Nb + Hf/Y wt.% base**
  - ~850-900°C temperature limit for Al<sub>2</sub>O<sub>3</sub> formation
- **Low Nickel AFA<sup>LN</sup> Grade : Fe-12Ni-14Cr-2.5Al-0.6Nb-5Mn-3Cu wt.% base**
  - ~650°C temperature limit for Al<sub>2</sub>O<sub>3</sub> formation

**Temperature Limit Based on Oxidation in Air + 10% H<sub>2</sub>O (~100°C)  
Higher-Temperature Oxidation Limit in "Dry" Air**

# Trial Heat of AFA Alloy Readily Welded

Gas Tungsten Arc Weld  
(used same alloy as a filler material)



- No crack appears at fusion/heat-affected zones

# Thermodynamic Calculation Results (750°C)

Alloy	750°C composition in austenite phase					750°C Calculated Phase Vol.%		
	Cr	Al	Ni	Nb	C	MC	Fe <sub>2</sub> Nb	NiAl
2-0.2	13.67	2.23	20.28	0.008	0.0618	0.07	1.66	1.49
2-0.9	14.52	2.11	19.64	0.012	0.0179	0.66	2.14	2.7
<b>3-0.4</b>	14.55	2.21	19.24	0.005	0.0384	0.4	2.24	4.71
<b>3-0.6</b>	14.53	2.21	19.21	0.006	0.0337	0.59	2.29	4.65
<b>3-1</b>	14.86	2.15	19.10	0.008	0.0199	0.76	2.53	4.97
<b>3-1.5</b>	15.38	1.91	18.42	0.026	0.0065	0.79	3.09	6.82
<b>3-2.5</b>	15.72	1.67	18.29	0.064	0.003	0.78	4.58	8.25
4-1 <sup>LNi</sup>	12.91	2.20	16.60	0.005	0.0188	0.64	2.4	10.68

Nb in the alloys  
(wt%)

← **0.4**

← **0.6**

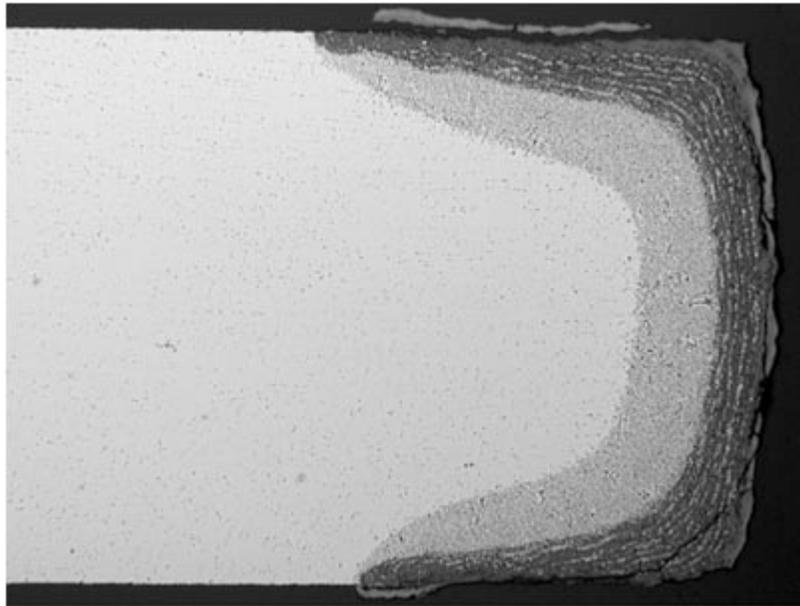
← **1.0**

← **1.5**

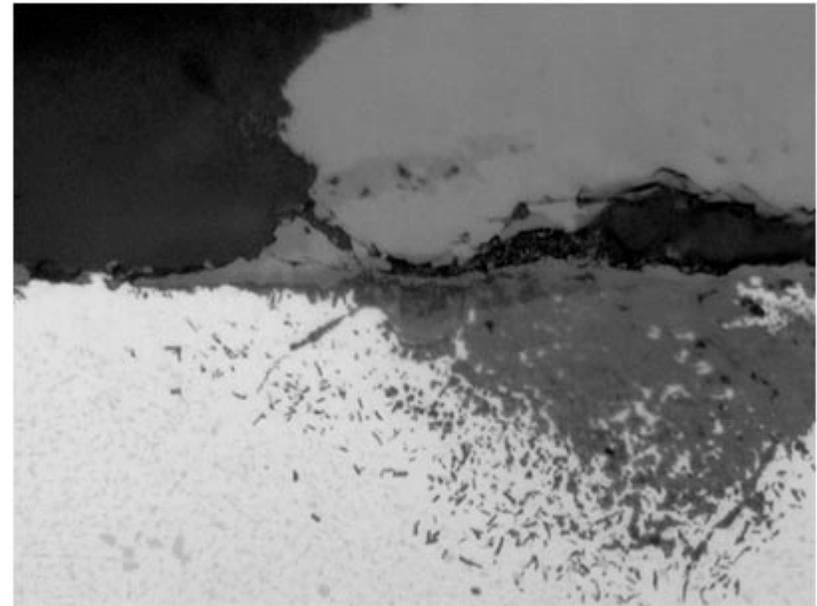
← **2.5**

# Fe-Rich Oxide Nodules + Extensive Internal Attack of Al When AFA Alloys Go "Bad" in Air + H<sub>2</sub>O

Optical Cross-Section of Fe-20Ni-14Cr-3Al-1.5Nb wt.% base after 1600 h at 800°C in Air + 10% Water Vapor



08-0917-01 MB 276 HTUPS 10  
800°C, H<sub>2</sub>O, 1600h 50µm  
As polished

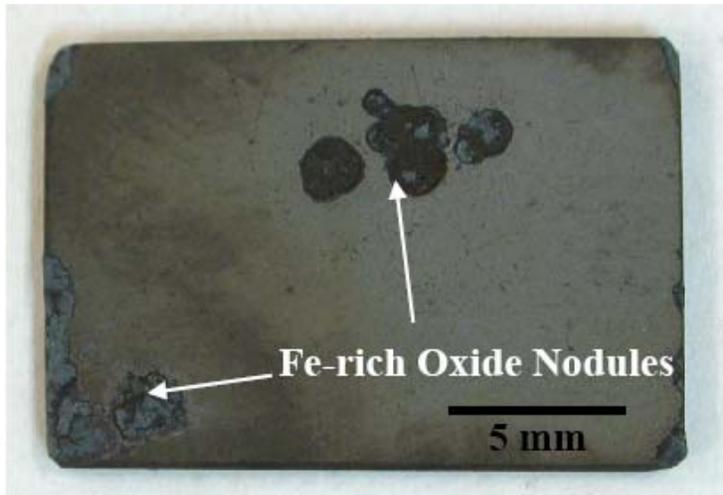


08-0917-03 MB 276 HTUPS 10  
800°C, H<sub>2</sub>O, 1600h 5µm  
As polished

- Raising temperature/adding water vapor favors transition to internal attack (transition temperature varies with composition)
- AFA alloys near borderline for Al<sub>2</sub>O<sub>3</sub> formation to co-optimize mechanical properties

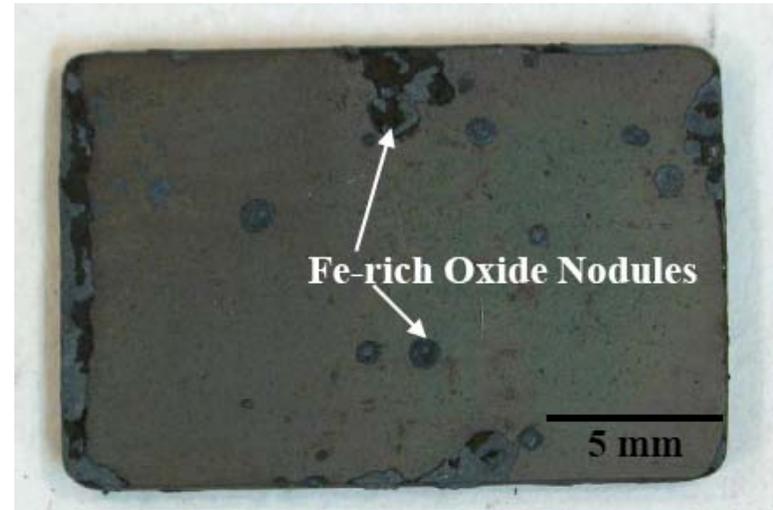
# After Cyclic Oxidation at 800°C in Air + H<sub>2</sub>O

**3Al-1.5Nb-14Cr**



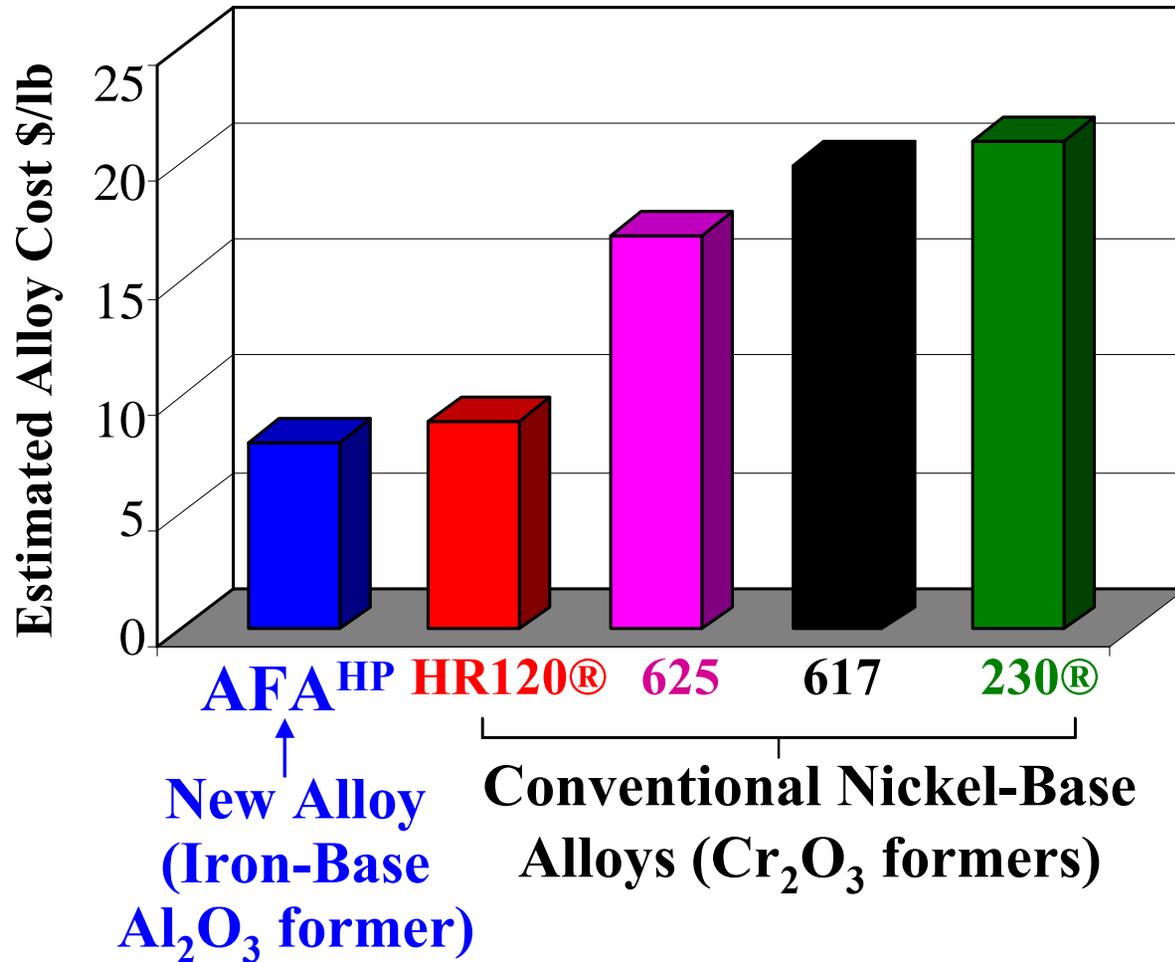
**(1600h)**

**3Al-2.5Nb-14Cr**



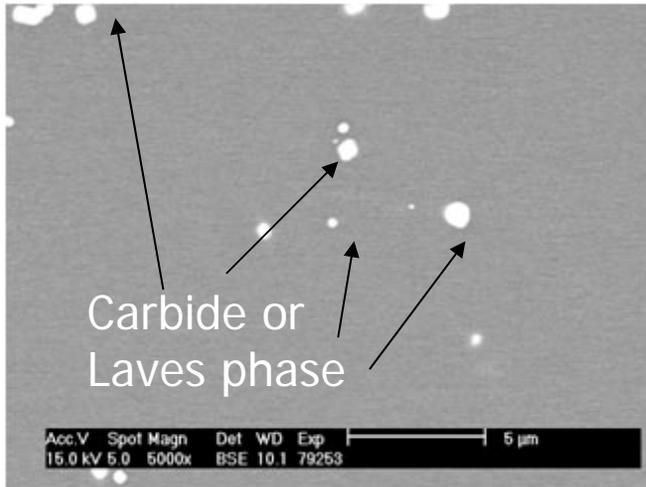
**(5700h)**

# AFA<sup>HP</sup> Significantly Lower Raw Material Cost to High-Ni Austenitics/Ni-Base Alloys

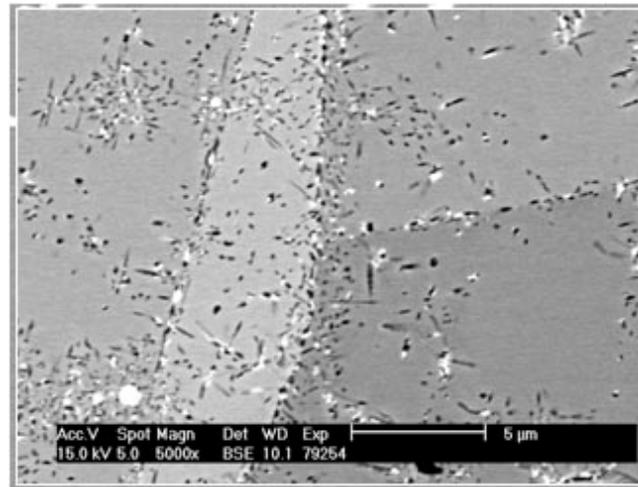


- AFA<sup>HP</sup> grade estimated comparable raw material cost to state-of-art austenitics such as alloy 709 (Fe-25Ni-20Cr base)

# B2 NiAl Precipitation on Ageing AFA 5 (Fe-20Ni-12Cr-4Al-1Nb wt.% base)

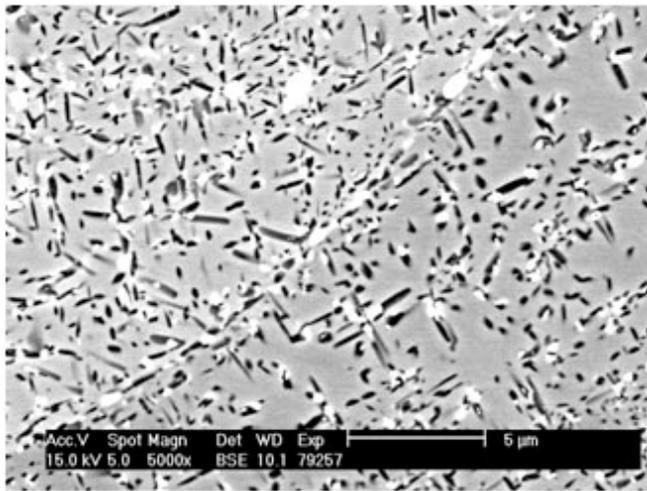


Before  
ageing

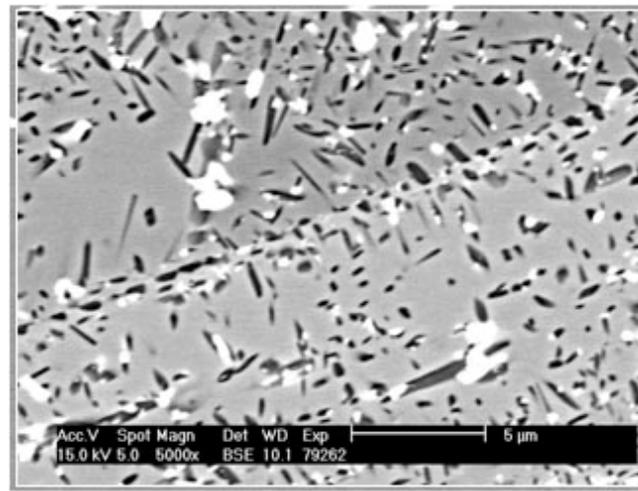


Ageing 50h

Black  
needle-like  
NiAl  
precipitates



ageing  
500h

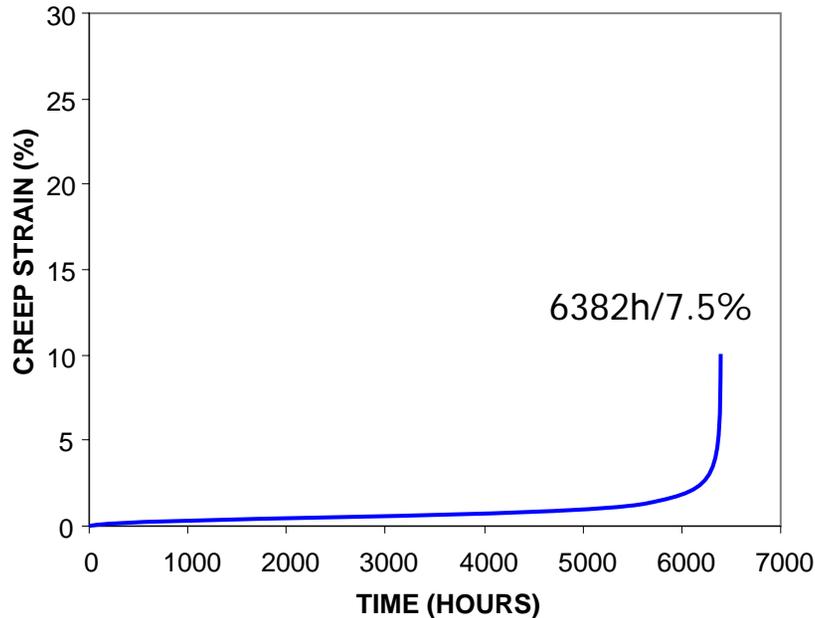


Ageing  
2012 h

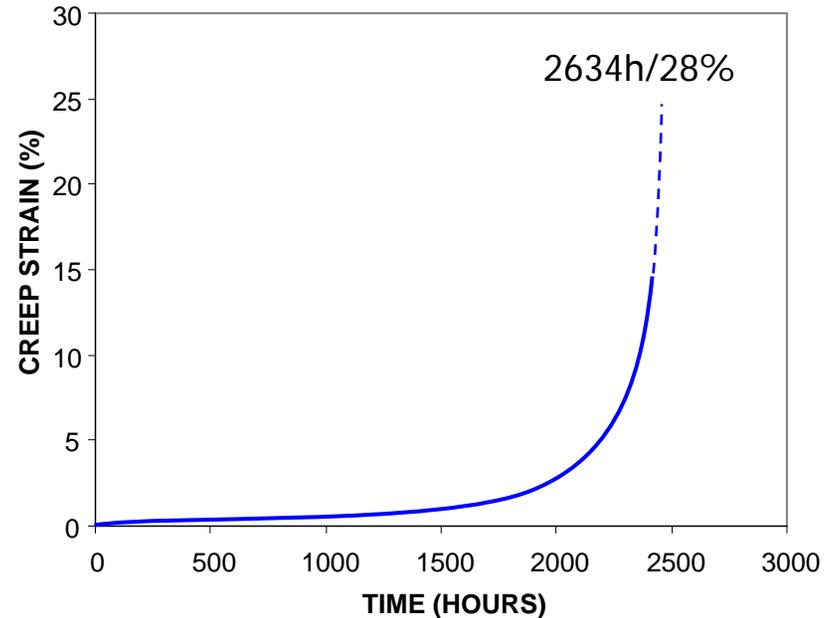
- NiAl precipitation after 50 h at 750°C
- Coarsening of NiAl reached mostly completed after ~500 h at 750°C

# Creep Curves Also Showed Necking Instability

(750°C/100MPa)



(800°C/70MPa)



- Acceleration creep started after only ~3% creep-deformation.
- Necking instability causes accelerating stress concentration.