# Advanced Measurement and Modeling Tools for Improved SOFC Cathodes

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Motivation: How do we identify and improve physical processes limiting cathode performance?



# The Problem:

# Current electrochemical techniques are limited in the information they can provide.

- Difficult to isolate the performance characteristics of the cathode from the rest of the cell under meaningful conditions.
  - Difficult to implement a reference electrode on a thin electrolyte.
  - Difficult to probe realistic current densities and polarization history.
- Limitations of impedance spectroscopy.
  - Overlap and ambiguity among physical processes (impedance is a "blob")
  - Lack of physical models linking characteristics to physics.
  - Uniqueness: an inherent limitation of linear response analysis.

# Our Approach

- Microelectrodes for improved cathode measurements
  - Better resolution and isolation than standard half-cells.
  - Allows testing of half cells under realistic bias conditions.
- Analysis of nonlinear harmonics (NLEIS, EFM)
  - Helps identify physical processes via. *nonlinearity*.
  - Broader spectrum of information without more experiments.
  - Stronger link between measurements and physical models.

# Materials of Interest

Porous Perovskite Electrodes:



Thin-Film and patterned electrodes:

La<sub>0.5</sub>Sr<sub>0.5</sub>CoO<sub>3-d</sub> on single-crystal YSZ



dense film of LSCF







## Purpose of a half-cell measurement



**Typical Experimental Goals:** 

- Measure voltage loss associated with a particular electrode in an operating fuel cell.
- Test electrode at a specific polarization (and history) in a particular environment.
- Isolate the electrode frequency response (impedance).

### Reference Electrodes are Prone to Error on Thin Cells



## **Error Includes Distortion of Impedance Spectra**



- "Cross-contamination" of WE and CE response.
- Distortion even with perfect electrode alignment.
- Impossible to avoid with thin cells having 1-D geometry.



# Theory behind a 2-D microelectrode



## Two approaches to 2-D "strip" microelectrode



# Verification of Microelectrode Half-cell Measurements



Procedure:

- 1) Measure V(i) characteristics of microelectrode.
- 2) Subtract *iR* losses (based on impedance).
- 3) Add anode and cathode overpotentials.
- 4) Compare to V(i)-iR for a fullsized symmetric cell having the same electrodes.

#### Conclusions:

- 2-D microelectrodes can be quantitative.
- For Pt, measured *iR* losses match calculation based on measured geometry.
- Accuracy requires good lateral current distribution (Pt).

# Microelectrodes not yet quantitative for perovskite electrodes



Porous La<sub>0.8</sub>Sr<sub>0.2</sub>CoO<sub>3-δ</sub> on SDC, 750°C, air Potentiodynamic Measurements





- Microelectrodes reveal qualitative differences b/t cathode and anode.
- Technique not yet quantitative due to poor lateral current distribution.

# **Current Status/Future Work**

#### Partial Mask



Full Mask



- To apply without correction factors, perovskites require a *full mask* config.
- It has proven difficult to produce thin, defect-free masks. Issues encountered:
  - mask porosity, thickness
  - expansion/adhesion of mask
  - electrolyte defects
- Current approach: 2 parallel paths
  - hybrid design with thick mask and metallic (Pt or Ag) bus.
  - thinner, screenprinted masks based on nanopowders.

What are harmonics, and why measure them?



- All nonlinear systems generate responses at multiples of the excitation frequency (harmonics).
- The magnitude, sign, and phase of the harmonics are highly sensitive to the details of the underlying physics.

#### Types on nonlinear harmonic measurements

Nonlinear Electrochemical Impedance Spectroscopy (NLEIS)

$$\omega_1 \longrightarrow \frac{\text{Process 1}}{\text{Process 2}} \longrightarrow \omega_{1,2}\omega_{1,3}\omega_{1,...}$$
 harmonic spectrum

Electrochemical Frequency Modulation (EFM)

$$\begin{split} & \omega_1 \longrightarrow \operatorname{Process} 1 \longrightarrow \omega_{1,2} \omega_{1,3} \omega_{1,\ldots} \\ & \uparrow & \operatorname{nonlinear}_{\text{interaction}} \longrightarrow \omega_1 + \omega_{2,2} \omega_1 - \omega_{2,\ldots} \\ & \omega_2 \longrightarrow \operatorname{Process} 2 \longrightarrow \omega_{2,2} \omega_{2,3} \omega_{2,\ldots} \\ \end{split}$$

Our apparatus for NLEIS and EFM measurements





## Voltage Signal Involves both Amplitude and Phase Shift





Power Series Expansion of Harmonic Response  

$$\overline{V}_{1}(\alpha, \omega_{0}) = \alpha \overline{V}_{1,1}(\omega_{0}) + \alpha^{3} \overline{V}_{1,3}(\omega_{0}) + \alpha^{5} \overline{V}_{1,5}(\omega_{0})$$

$$\overline{V}_{3}(\alpha, \omega_{0}) = \alpha^{3} \overline{V}_{3,3}(\omega_{0}) + \alpha^{5} \overline{V}_{3,5}(\omega_{0})$$



Harmonic Spectra  $\overline{V}_{k,l}(\omega)$  of La<sub>0.8</sub>Sr<sub>0.2</sub>CoO<sub>3- $\delta$ </sub> on SDC at 750°C in air



# Extending existing models to the nonlinear regime



# Extending existing models to the nonlinear regime

- Questions to be tested using the model:
  - Does the nonlinear response of LSC on SDC obey a bulk pathway model?
  - Is the solid-solid interface at equilibrium, as implied by the linear response?
  - What is the rate expression and mechanism of oxygen ad(b)sorption?
- Modeling assumptions:
  - 1-D geometry with average microstructural parameters.
  - Nonstoichiometry and diffusion based on bulk materials properties.
  - Interfacial charge-transfer based on Butler-Volmer rate equations.
  - O<sub>2</sub> reduction involves a single rate-determining step:

$$r = k \left( \frac{\left( P_{O_2}^{gas} \right)^n}{\left( P_{O_2}^{Solid} \right)^m} - \frac{\left( P_{O_2}^{Solid} \right)^{n-m+q}}{\left( P_{O_2}^{gas} \right)^q} \right)$$

- Solution method:
  - Dual expansion of governing equations in a Fourier/power series.
  - Numerical formulation and solution using finite element analysis (FEMLAB).

#### Adsorption-limited O<sub>2</sub> exchange

 $rds: O_2^{(gas)} + 2s \leftrightarrow 2O^{(ads)}$ 

$$r = k \left( \frac{\left(P_{O_2}^{gas}\right)^1}{\left(P_{O_2}^{Solid}\right)^{1-n}} - \left(P_{O_2}^{Solid}\right)^n\right)$$
(shown:  $n = 0.5$ )

- Model consistent with measured 1st harmonic (Gerischer), including -45° phase angle at high frequencies.
- Loss of higher harmonics (linearity) at high frequency confirms absence of polarization at the solid-solid interface.
- Sign and magnitude of harmonics at low frequency are not consistent with adsorption-limited O<sub>2</sub> exchange.



Incorporation-limited O<sub>2</sub> exchange

$$rds: O^{(ads)} + V_O^{\bullet \bullet} + 2e^- \leftrightarrow 0$$

$$r = k \left( \left( P_{O_2}^{gas} \right)^n - \left( P_{O_2}^{Solid} \right)^n \right)$$
  
(shown:  $n = 0.5$ )

- Model degenerate with previous model for the first harmonic.
- Symmetric rate expression results in nullifications (sign flips) in the higher harmonics, not present in the data.
- Sign and magnitude of harmonics at low frequency are not consistent with incorporation-limited O<sub>2</sub> exchange.



**Empirically-derived rate** 

$$r = k \left( \frac{\left( P_{O_2}^{gas} \right)^1}{\left( P_{O_2}^{Solid} \right)^{93}} - \frac{\left( P_{O_2}^{Solid} \right)^{1.5}}{\left( P_{O_2}^{gas} \right)^{1.47}} \right)$$
  
rds: (none possible)

*rds*: (none possible)

- Model degenerate with other models for the first harmonic; but better agreement in all harmonics over entire spectrum.
- Predicted weak exchange rate ( $\sim P_{O_2}^{0.1}$ ) is consistent with weak dependence previously observed in  $r_0 vs P_{O2}$ .
- This rate expression very hard to justify based on a single rate-determining step, plus bulk diffusion (only).



# Conclusions

- Higher harmonics provide additional nonlinear information that *severely restricts* possible mechanisms underlying a.c. behavior.
- When fired at sufficiently high temperatures, LSC on SDC exhibits nearly zero resistance to charge-transfer at the solid-solid interface.
   Polarization is due to chemical dissociation and transport.
- For La<sub>0.8</sub>Sr<sub>0.2</sub>CoO<sub>3-δ</sub> at 750°C in air, higher harmonics do not seem consistent with a bulk path involving a single *rds* for O<sub>2</sub> exchange. Possible explanations:
  - multiple rate-determining steps for  $O_2$  exchange.
  - local morphological effects.
  - parallel surface path (favored explanation).

# Current/Future Work (NLEIS)

- NLEIS measurements on dense thin-film perovskite electrodes, eliminating surface path.
- Measurements of LSCF porous electrodes over a wider range of temperature,  $P_{O2}$ , Sr content, surface area, and interfacial bonding.
- Measurements of 2nd and 4th harmonics (half cells).



- Microelectrodes potentially offer an easy, reliable vehicle for half-cell measurement and testing.
- NLEIS shows promise as a superior method for electrode analysis and characterization, with little additional experimental effort/time vs. EIS.