# Low Thermal Conductivity, High Durability Thermal Barrier Coatings for IGCC Engines 10/1/11

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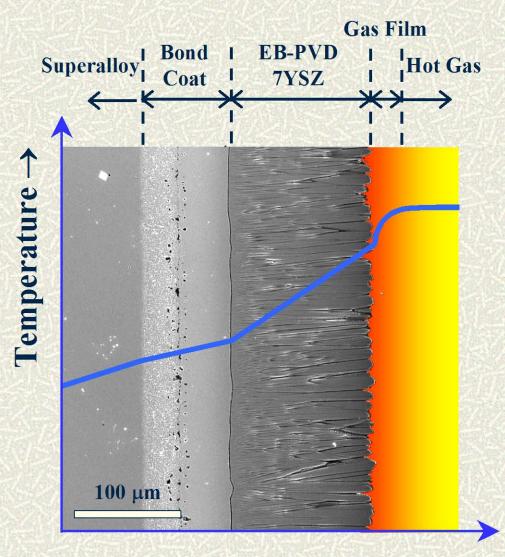
University of Connecticut

Dr. Briggs White DOE Program Manager DE-SC0006814

## **Critical Industrial Participation**

- Pratt Whitney and Siemens will supply bond coated superalloy substrates.
- Pratt Whitney will make CMAS glass for use in CMAS exposure testing. (and advise on composition of the glass)
- Pratt Whitney is supplying on Ph D employee student to work on the project

#### Microstructure & Requirements For TBCs



#### **TBC Requirements**

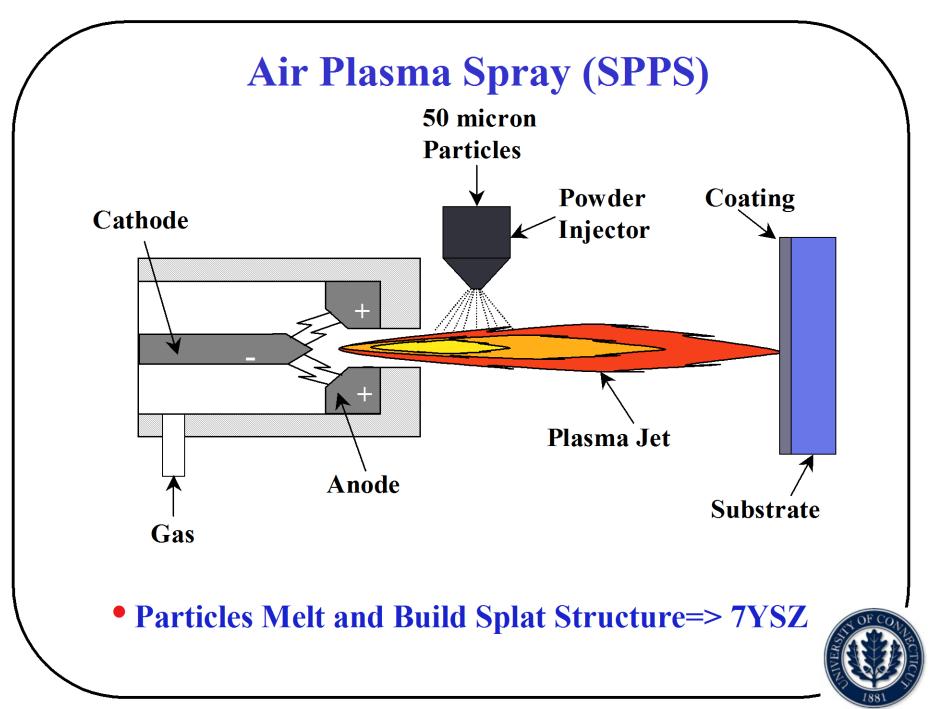
- Low Thermal Conductivity
- High Durability
  - > Toughness
  - > Strain Tolerance

#### Goals

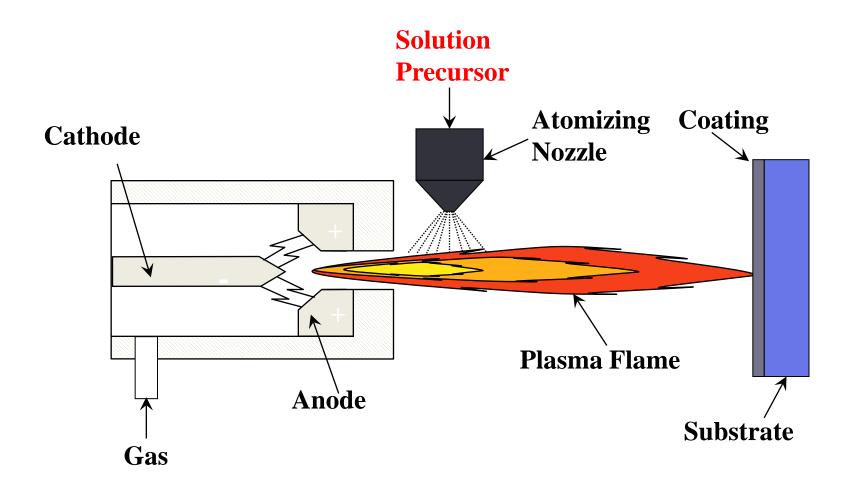
- Reduce thermal conductivity of YSZ TBCs to 0.5 watt/m°K by use of inter-pass boundaries (IPBs).
- Increase the allowable surface temperature of the YSZ TBCs by 100°C by use of thin, high temp. surface corrosion barriers layers (CBLs).
- Improve TBC durability in CMAS environment by use of CBLs + other methods.
- Reduce the use of rare earth elements compared to other low K TBCs

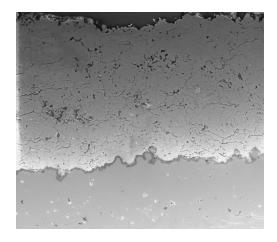
# **Key Program Processes/Tests**

- UConn Thermal Spray Facility
- Solution Precursor Plasma Spray Process (SPPS)
- TBC Cyclic Furnace Testing Facility
- Moist Environment Testing (being build for this program)



#### **Solution Precursor Plasma Spray Process**





#### APS

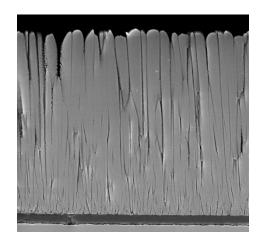
Lower Thermal Cond.



Shorter Spallation Life



Maxim thickness 1 mm ( $\sqrt{}$ )



#### **EB-PVD**

Higher Thermal Cond.

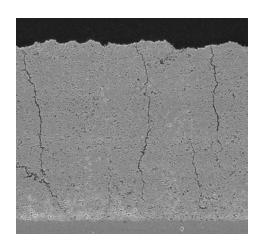


Longer Spallation Life



Maxim thickness 0.25

mm (**X**)



#### **SPPS**

Lower Thermal Cond.



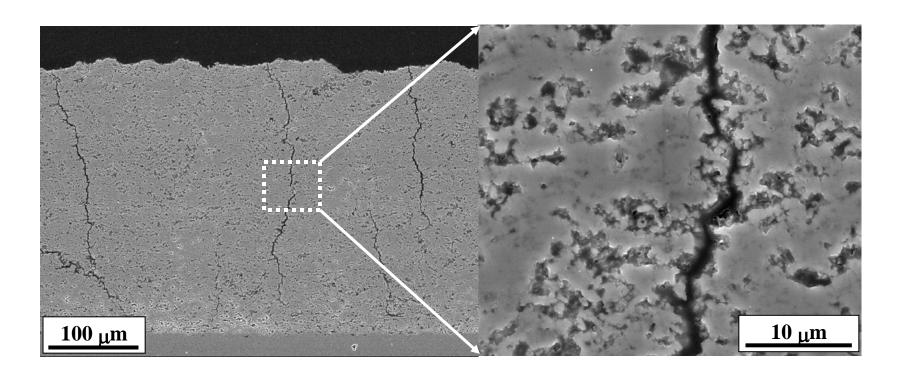
Longer Spallation Life



Maxim thickness 2.5



#### **Microstructure Of SPPS TBCs**



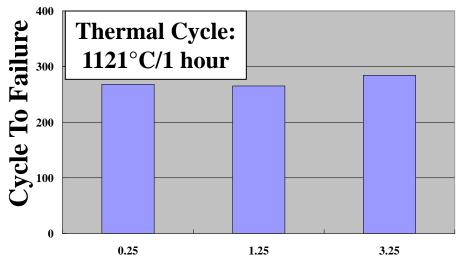
#### **Unique Features**

- 3D Nano & Micrometer Porosity
- Through-Thickness Cracks
- Ultra-Fine Splats

## Advantages of Solution Spray

- Vertical stress relieving Cracks- Critical when thick
- Higher Fracture Toughness
- Rapid Composition Exploration (100X)
- Structured Porosity leading to low K coatings

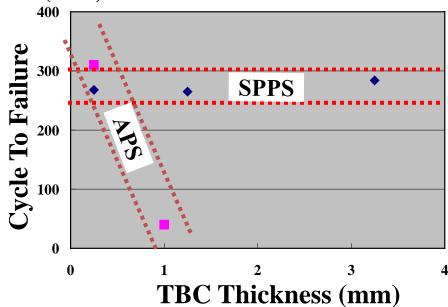
#### **Effect of SPPS TBC Thickness On Durability**



**Substrate: CMSX-4** 

**Bond Coat: Co210** 





# SPPS Coating have 5X higher In Plane Toughness

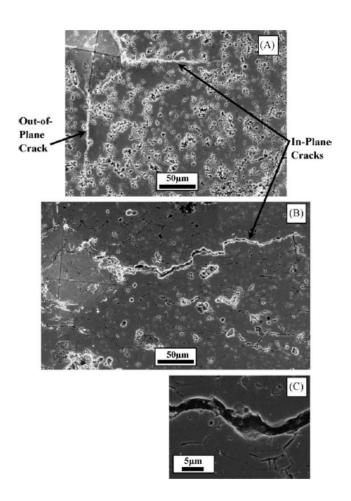
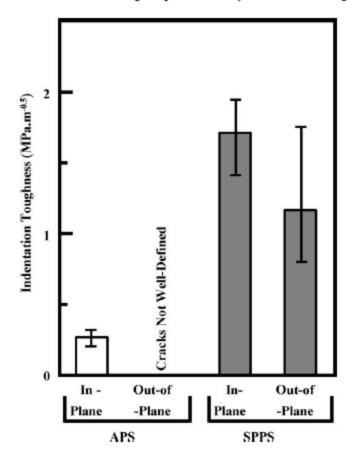
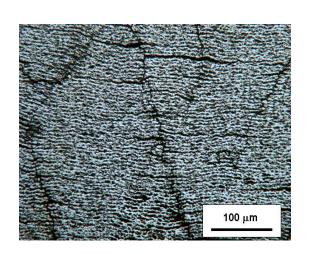
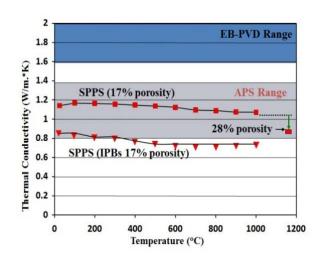


Fig. 4. SEM micrographs of Vickers indentation (49 N load) sites in: (A) SPPS coating and (B) APS coating. Arrows indicate in-plane and out-of-plane cracks. (C) High-magnification SEM micrograph showing the in-plane indentation crack following a "splat" boundary in the APS coating.



# Structured Porosity Lead to Lower Thermal Conductivity

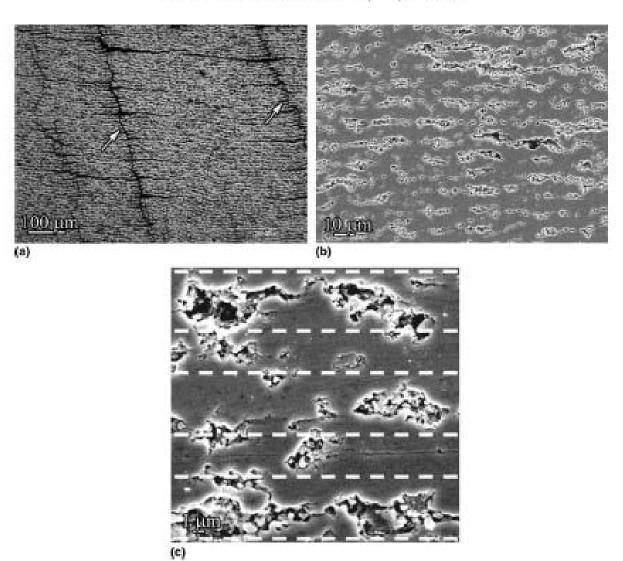




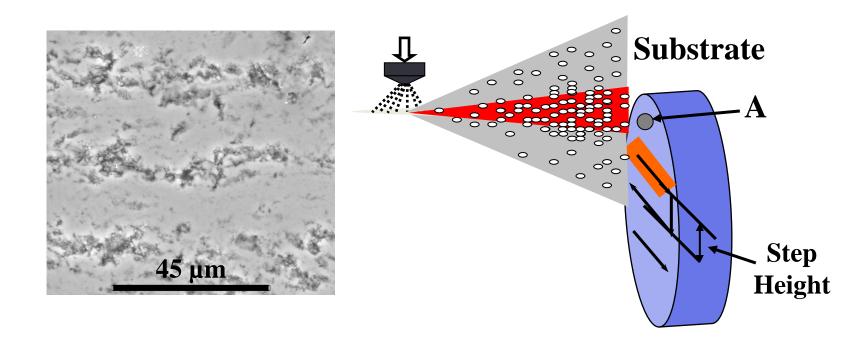
**Figure 3.** (a) SPPS YSZ TBC with IPBs (a) and resultant, reduced thermal conductivity

### **IPB** Microstructure

A.D. Jadhav et al. / Acta Materialia 54 (2006) 3343-3349



#### **Formation of Inter-Pass Boundaries**



# **UConn Thermal Spray Facility**



#### **Metco 9MB Plasma Spray System**



**Metco 9MC Control** 





9MB Plasma Gun

4MP Dual Powder Feeder

# **Liquid Delivery Options**



**Standard Inframat System** 



**Unique High Pressure System (33 atm)** 

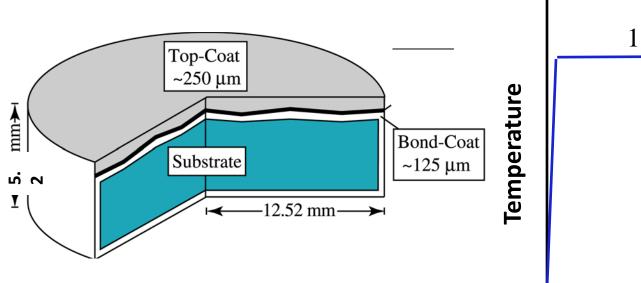
# Cyclic Furnace Test Facility



#### **Specimen Shape & Furnace Cycle**

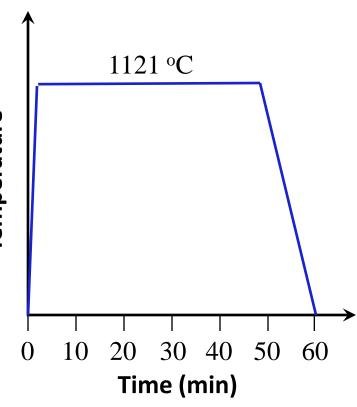
Disk-Shape Samples

Thermal Cycling Life Test





- Bond-Coat: APS NiCoCrAlY
- Top-Coat: SPPS Layered



#### Vast Reference Data Base

- GTD 111- GE DV
- HY 230 with Ni Co Cr Al Y and SPPS
- Rene N-5 Pt-Al BC with EB-PVD
- CMSX-4, Pt Al with EB-PVD
- CMSX-4, MCrAlY+Si, Hf-EB-PVD
- PWA 1484- APS Ni Cr Al Y -APS

# 10 More Types Tested

#	S1	S2	S3	S4	S5	S6	S7	S8	A1	E1
Top Coat Process	SPPS	SPPS	DVC/APS	EB-PVD						
Bond Coat Category	MCrAlY	MCrAlY	MCrAlY	Pt-Al						
Bond Coat Process	APS	APS	APS	HVOF	HVOF	LPPS	LPPS	LPPS	APS	CVD
Substrate	H230	H230	NA	H230	CMSX-4	CMSX-4	MARM509	NA	GTD-111	CMSX-4

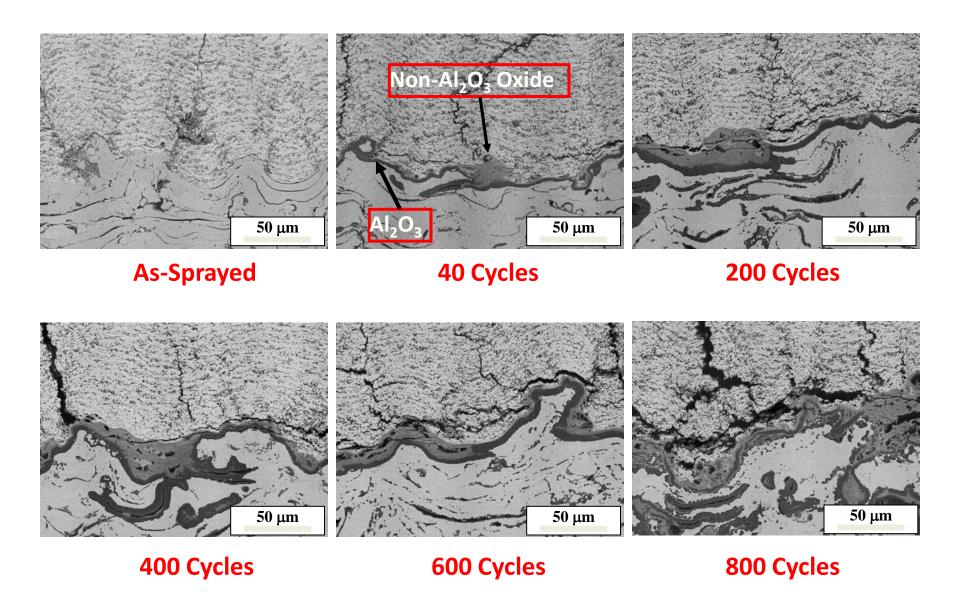
# **Bond Coat Composition**

	Со	Ni	Cr	Al	Y	Si	Hf
S1, S2		bal	20	10	1		
S4	23-26	Bal.	15-19	9-11	0.2-0.4		
S5, S6	Bal	32	21	8	0.5		
S7	20	Bal	18	12.5	0.6	0.4	0.25
?\$8	Bal	32	21	8	0.5		

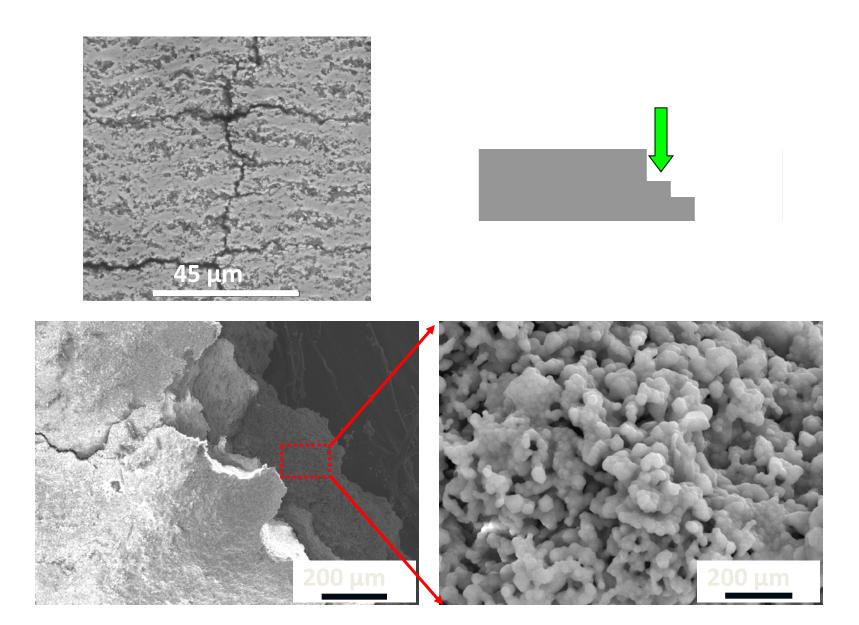
## Test Results at 1121 C 1 Hour Cycles

#	S1	S2	S3	S5	S6	S7	S8	A1	E2
Cycle Life	1140 1140* 1230 1230	573, 595* 785* 1049* 1206	697*	300* 350	245, 270* 290	220, 245*	125 140*	Ave. 300	Ave. 458
Failure % in ceramic	70%	60-75%	50%	60%	25%	50%	30%	-	-

#### **Change of Microstructure with Thermal Cycling**



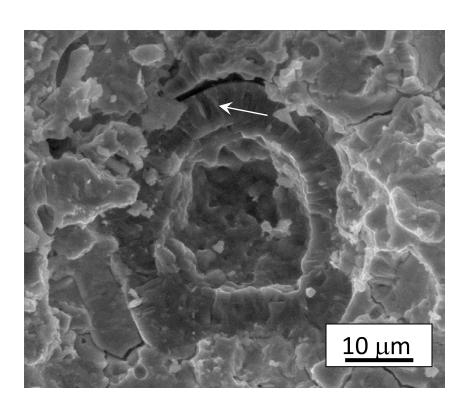
#### **Morphology of Inter-Pass Boundary**

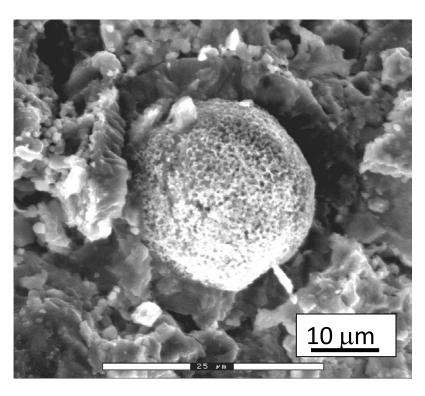


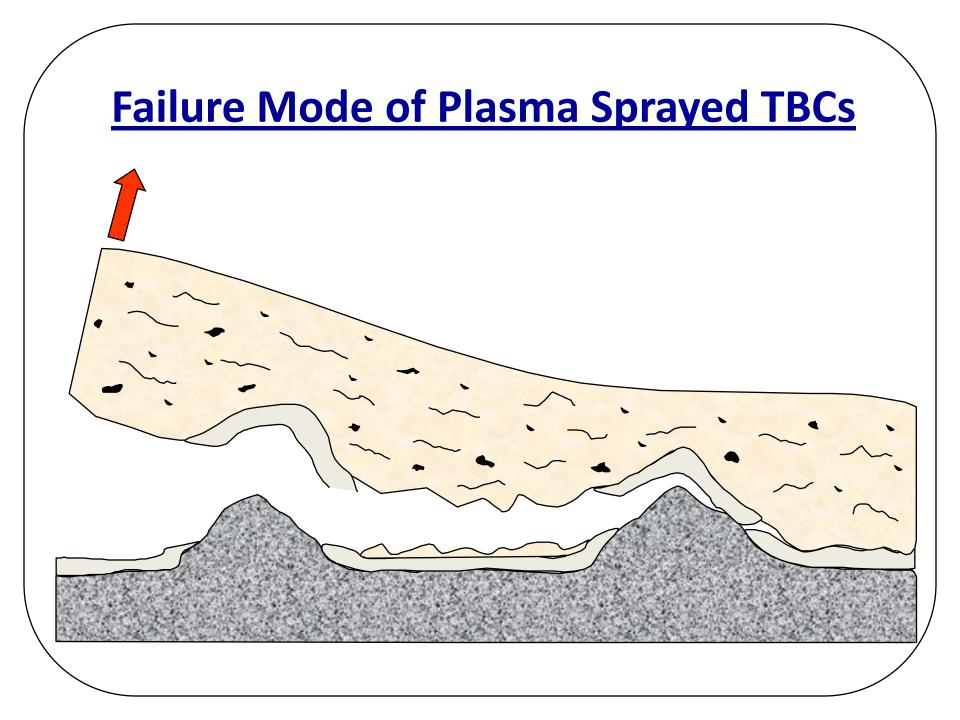
### **Spallation Surface Features**

Ceramic "Bottom" Side

• Metal "Top" Side





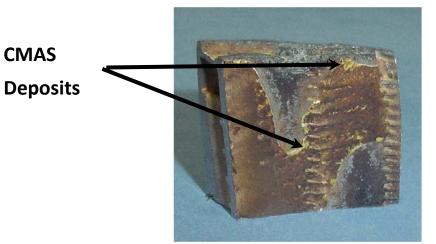


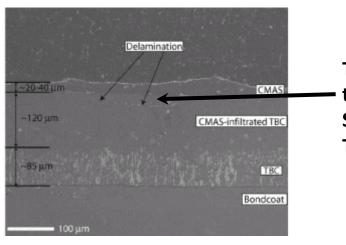
#### **Contaminants Affect Failure**

Calcium, Magnesium, Aluminum Silicon= CMAS

#### CIMAS Infiltration of 7YSZ Thermal Barrier Coating

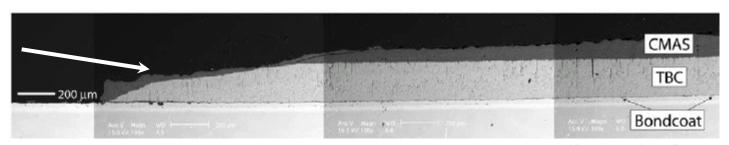
#### **Field Observation of CMAS Attack**





Transverse Cracks that Lead to Shedding of Topcoat

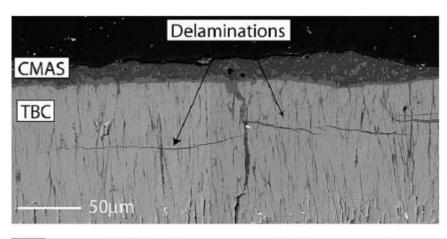
Coating Loss
Due to CMAS
Infiltration



Mercer et al. 2005

# 1. Loss of Strain Tolerance-Mechanical Effect

A.G. Evans, J.W. Hutchinson / Surface & Coatings Technology 201 (2007) 7905-7916



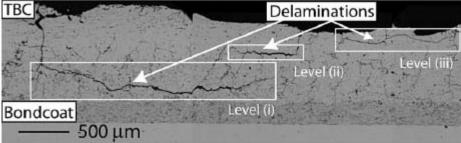


Fig. 1. Examples of delaminations in thermal barrier coatings obtained from components removed from engines subjected to CMAS penetration: (a) Sub-surface mode I delaminations in an airfoil with a TBC made by electron beam physical vapor deposition; the delaminations are within the penetrated zone [9]. (b) Delaminations at several locations within a shroud penetrated by CMAS; the TBC is 1 mm thick and deposited by air plasma spray (APS) [10].

# Mechanics Modes for Loss of Strain Tolerance Developed by Hutchinson and Evans

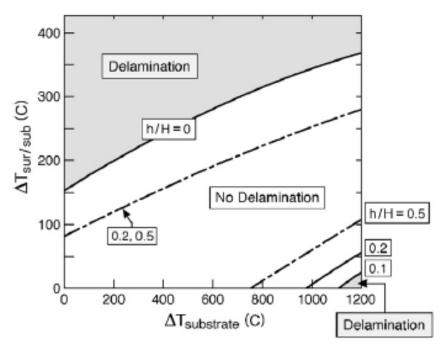


Fig. 10. A map for deep delamination in an APS-TBC on a superalloy substrate with CMAS infiltration to depth, h/H. The mixed mode toughness parameter is,  $\lambda$ =0.25.

# 2. Many types of chemical and Phase Effects for example Y loss and destabilization of t phase $Zr\ O_2$ to Monoclinic with a destructive volume change

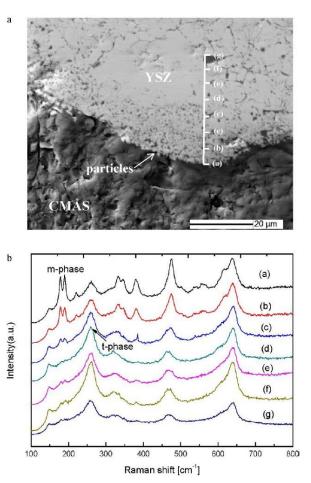


Fig. 4. (a) Micrograph of the interaction zone of CMAS deposit and YSZ coating after 4 h heat-treatment at 1250 °C, and (b) Raman spectra obtained from the positions marked in (a).

# **Proposed Work Plan**

# **Program Plan**

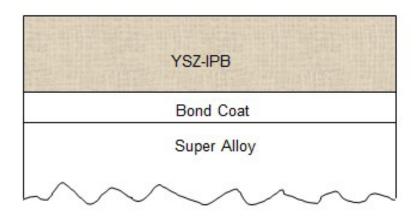
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	YEAR 1			YEAR 2				YEAR 3				
TASK	Q1	Q2	Q3	Q4	Q1	Q2	Q3	Q4	Q1	Q2	Q3	Q4
1.Manage/Plan/Report												
2. SPPS of Low K TBC												
3. Test Low K TBC												
4. Fabricate Gd-Zr Layer												
5. Fab. Al-Ti Doped TBC												
6.Fab. CaSo₄ Additive												
7. CMAS Testing												
8. Moisture Testing												
9. Define Mechanisms												
10 Gradient Cyclic												
Testing												

## Minimization of Thermal Conductivity Using IPBs

- Experimental Design of spray trials
- Outcome assessment by thermal conductivity calculation verified by measurement

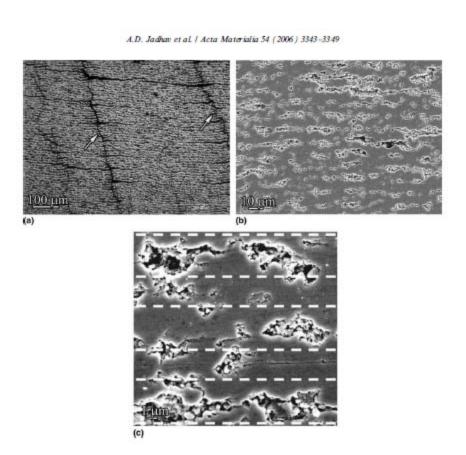
#### Base Line System



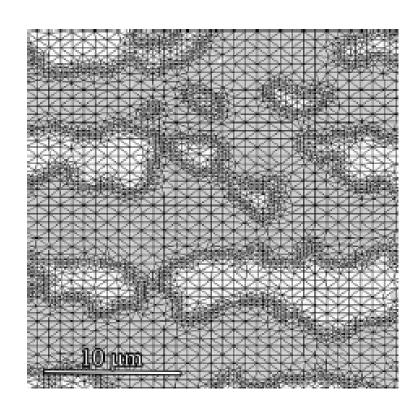
**Figure 6.** TBC #1, a Low K SPPS YSZ TBC using IPBs and porosity

### Rapid Measurement of Thermal Conductivity

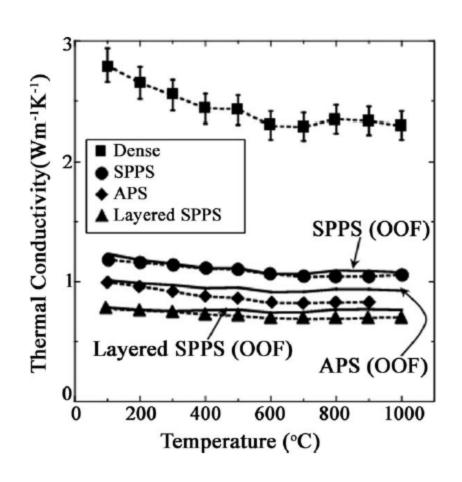
### Calculating Thermal Conductivity



# Finite Element Mesh Generated from Micrograph Using OOF Program



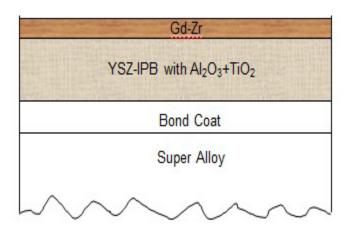
## Calculated Conductivity Agrees Well with Experiments



### CMAS Damage Mitigation to be Implemented

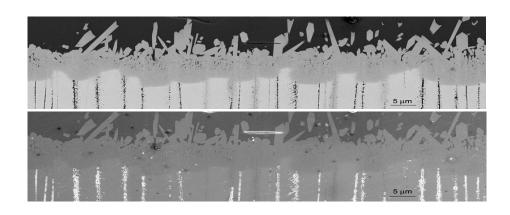
Three Approaches

### 1. Add Gd-Zr to baseline system for higher temperature phase stability and CMAS



**Figure 7.** TBC system #2 with low conductivity solution plasma sprayed YSZ with IPBS and CMAS resistant high temperature tolerant Gc-Zr protective surface layer (PSL).

#### **CMAS** Resistance of GdZr



## Add Metastable Al<sub>2</sub>O<sub>3</sub> + to block CMAS in the YSZ layer

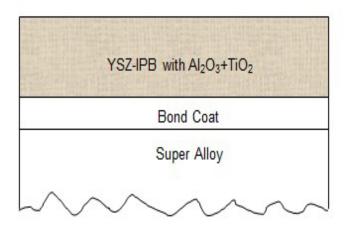
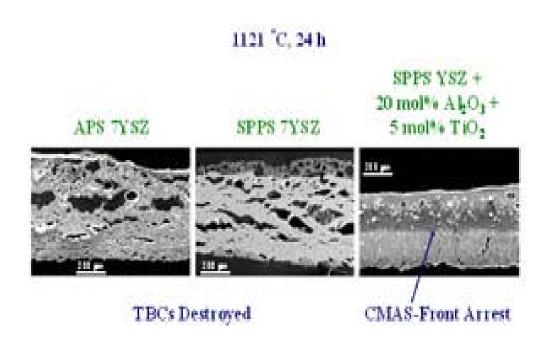


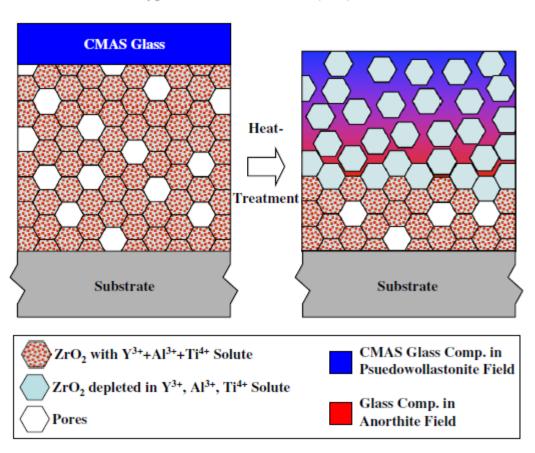
Figure 9. TBC system #4 has features of TBC #1-3 with calcium sulfate infiltration.

#### 2. Addition of metastable Al + more

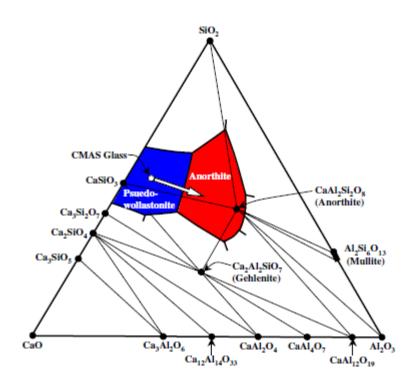


#### CMAS composition is altered to Anorthite Filed

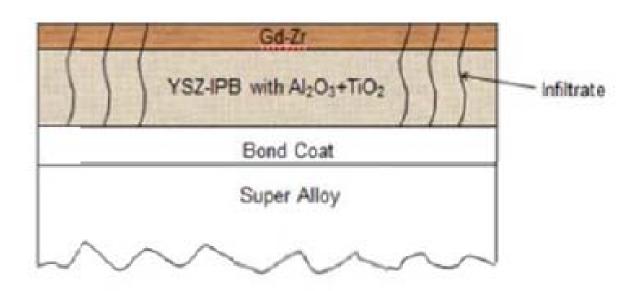
A. Aygun et al. | Acta Materialia 55 (2007) 6734-6745



# Microscopy Shows Anorthite phase is blocking

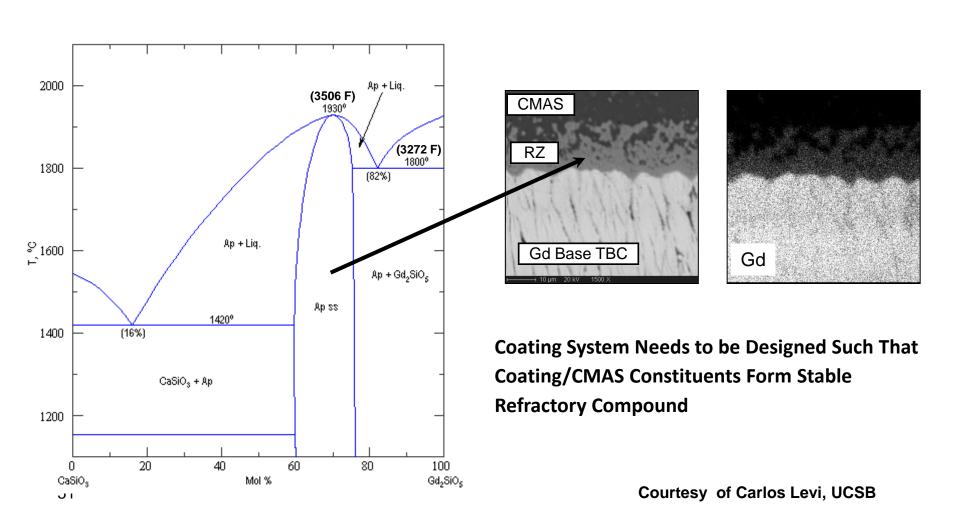


### 3. Infiltration of CaSO4 via a low melting eutectic of NaSo4-CaSo4-MgSo4



#### Analysis of Gd<sub>2</sub>Zr<sub>2</sub>O<sub>7</sub>/CMAS Reaction Product

#### Sealant Layer Identified as Hexagonal Apatite Phase, CaGd<sub>4</sub>(SiO<sub>4</sub>)<sub>3</sub>O



### 3. Infiltration with CaSO4 found in the field by Breau

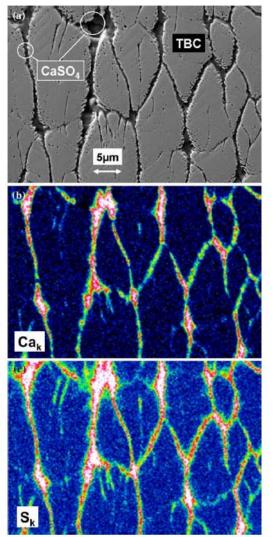


Fig. 3 a middle section of the YSZ top coat displaying  $CaSO_4$  infiltration of open porosity (suction-surface/region B, SEM, secondary electron image), b and c elementary mapping  $(Ca_k, S_k)$  proving that  $CaSO_4$  is continuous within the intercolumnar pore network of the coating

# Composition of the CMAS blocking Ca-Mg-Si materials found

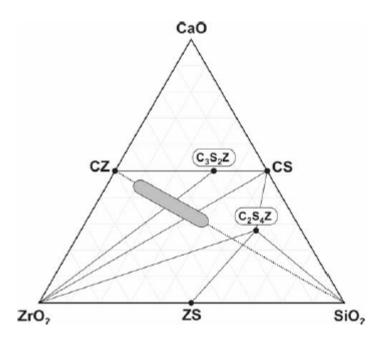


Fig. 13 Compatibility triangles in the ternary  $CaO-ZrO_2-SiO_2$  (mol%) at 1,300 °C (modified after [39, 40]) serving as a first approximation for pertinent phase assemblages at the CMAS/YSZ interface. Shaded area represents the compositional variation of the Ca-Zr-Fe-silicate which plots close to the join  $CZ-SiO_2$  (given by dotted line). Abbreviations: C=CaO,  $S=SiO_2$ ,  $Z=ZrO_2$ ,  $CZ=CaZrO_3$ ,  $C_3S_2Z=Ca_3ZrSi_2O_9$ ,  $C_2S_4Z=Ca_2ZrSi_4O_1$ ,  $CS=CaSiO_3$ ,  $CS=ZrSiO_4$ . Compatibility triangles in the subsystem CaO-CZ-CS are omitted

#### Summary

- Project Goals are;
  - Reduce thermal conductivity 0.5 watt/m°K
  - Increase surface temperature allowable by 100C
  - Significantly improve CMAS resistance
- Inter-Pass Boundaries will be used and optimized to lower thermal conductivity
- A top layer of GdZr will be used to:
  - Allow +100°C surface temperature + high purity SPPS
  - Reduce CMAS attack
- Al-Ti Metasable solutes+ will be added to the YSZ to reduce CMAS infiltration.
- CaSO4 will be used for the first time to arrest CMAS infiltration.

#### Summary

- Results will be validated with CMAS testing in cyclic furnace tests and high moisture tests
- Detailed mechanism of failure will be evaluated for the test run and Modification of the TBC will be made as mechanisms suggest.
- Substrates are to be supplied by Siemens and Pratt and Whitney

#### Questions?